

Supporting Information

Effect of building block connectivity and ion solvation on electrochemical stability and ionic conductivity in novel fluoroether electrolytes

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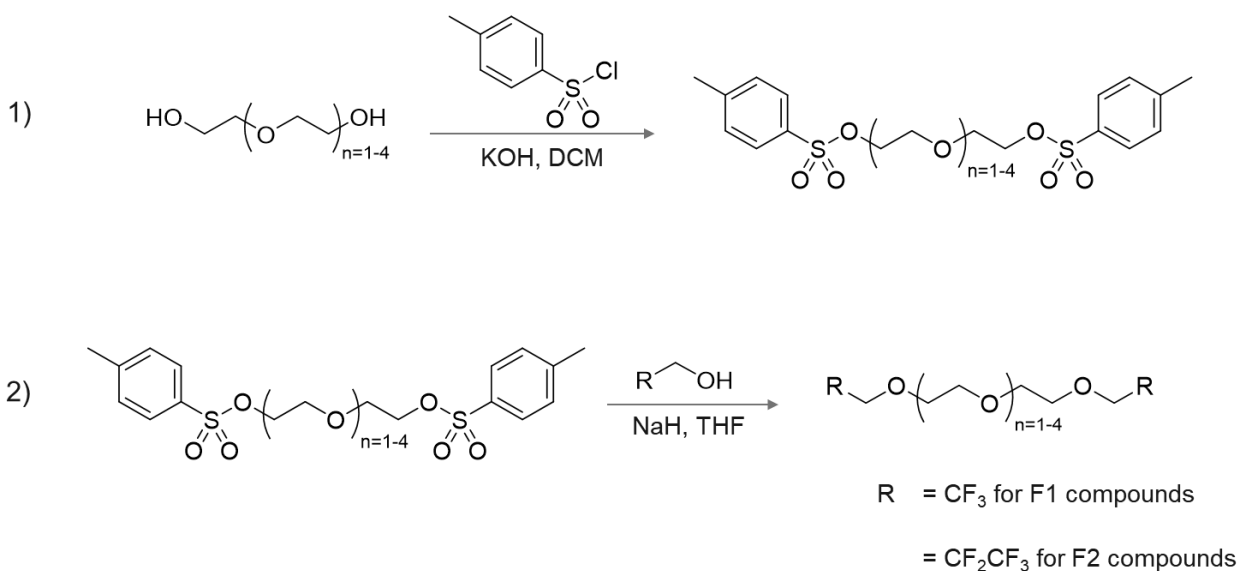
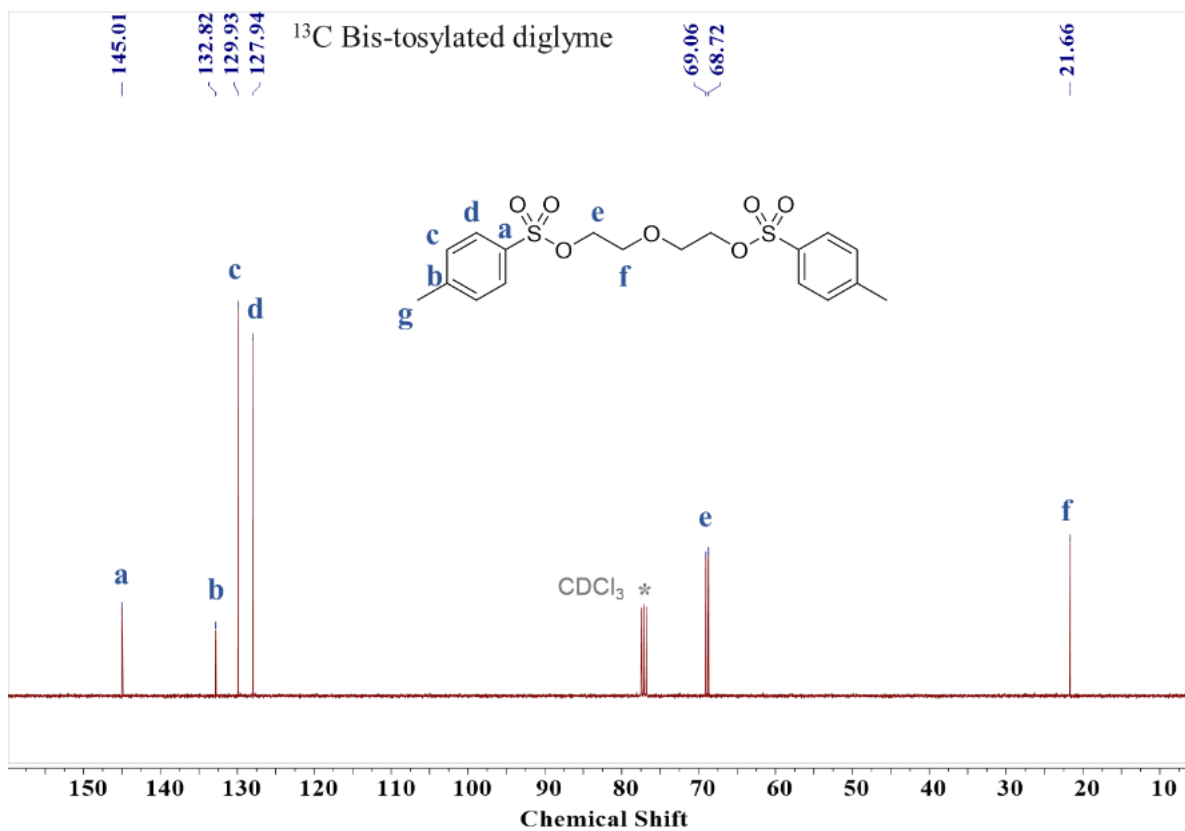
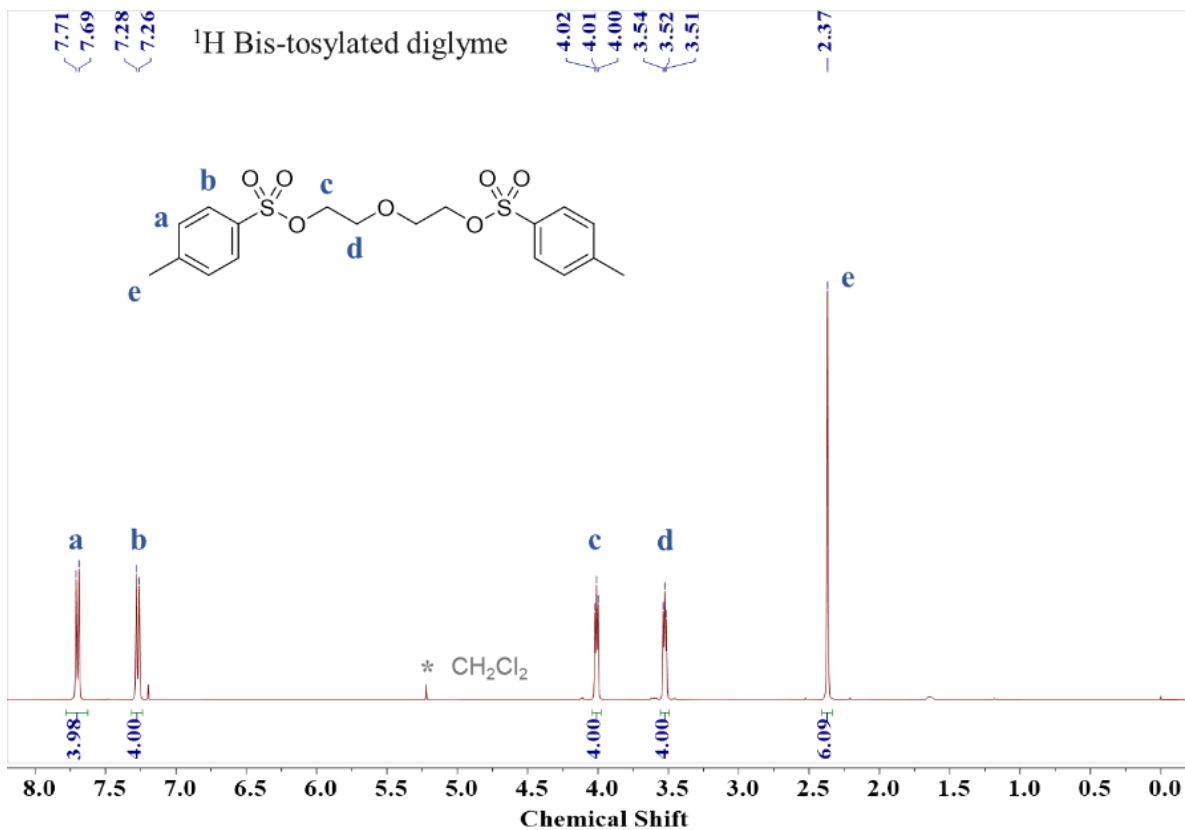
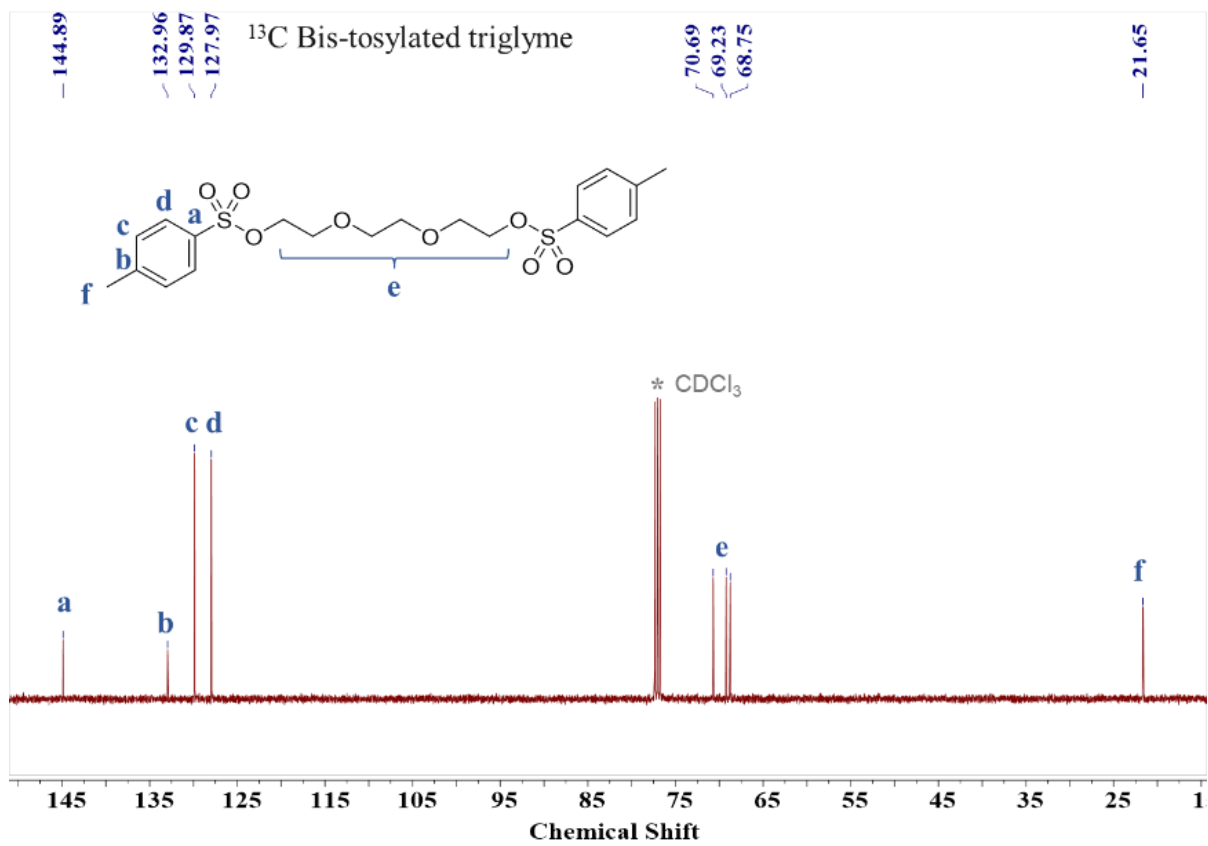
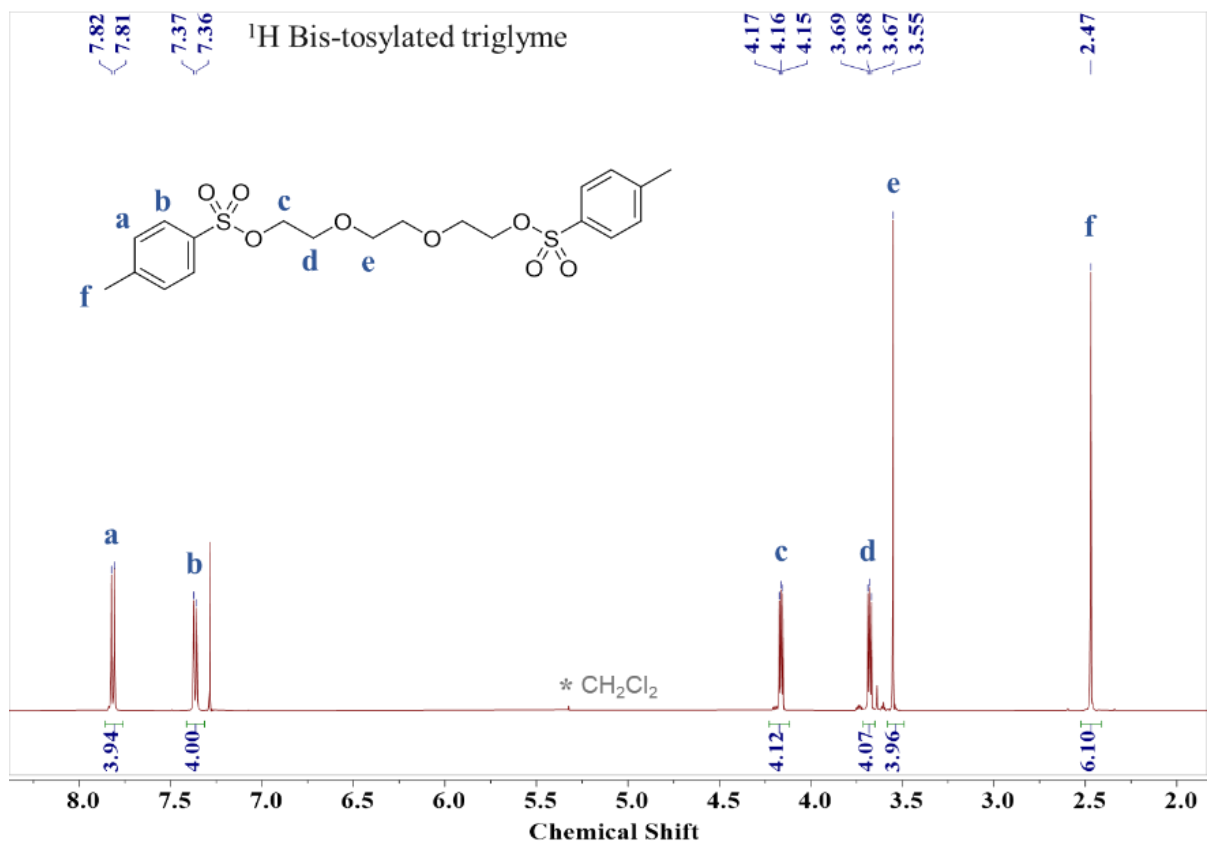
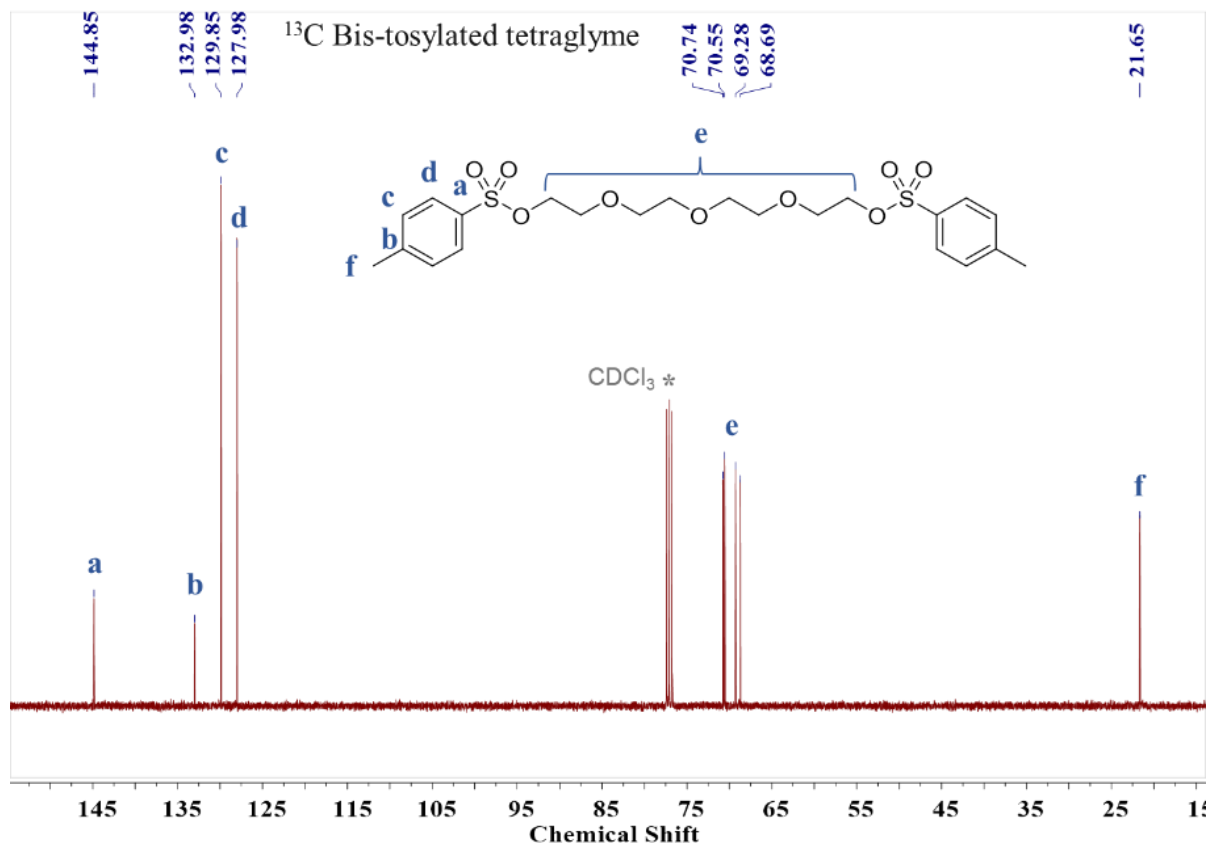
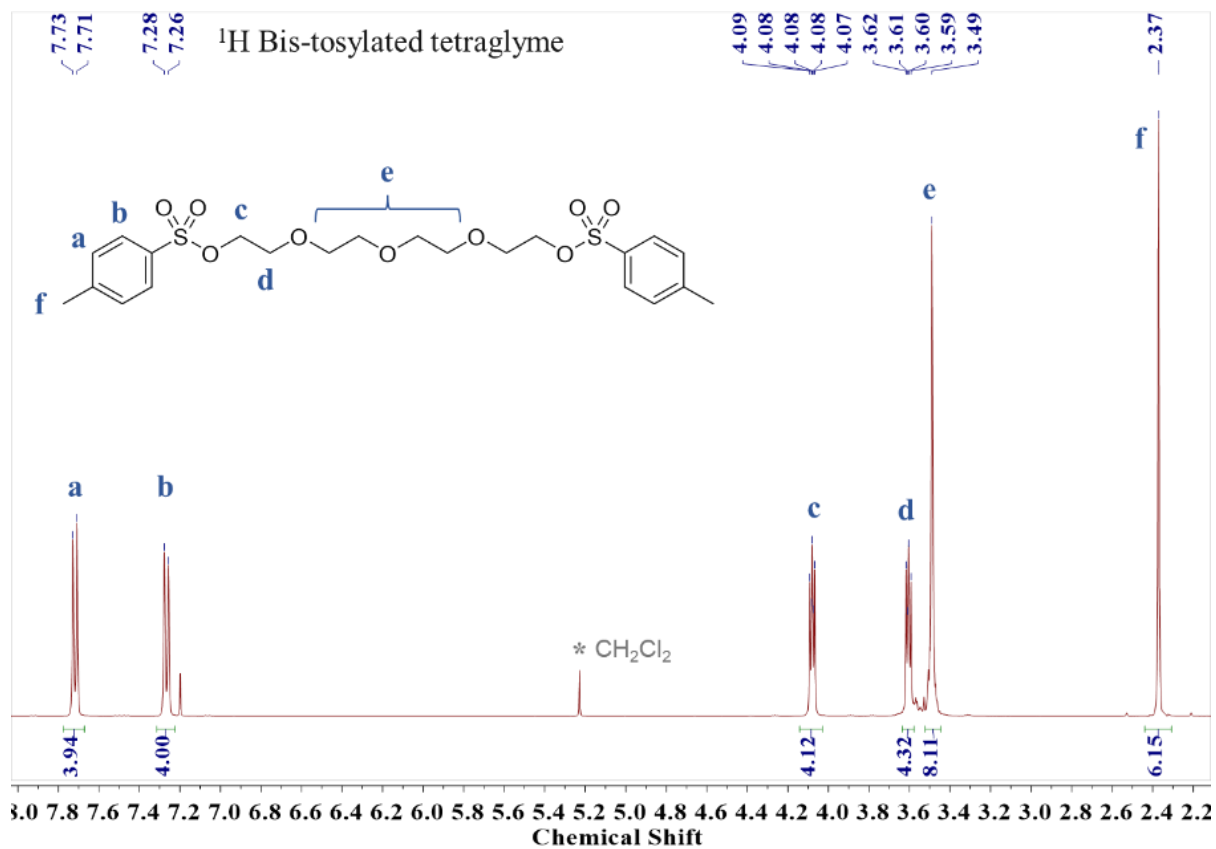
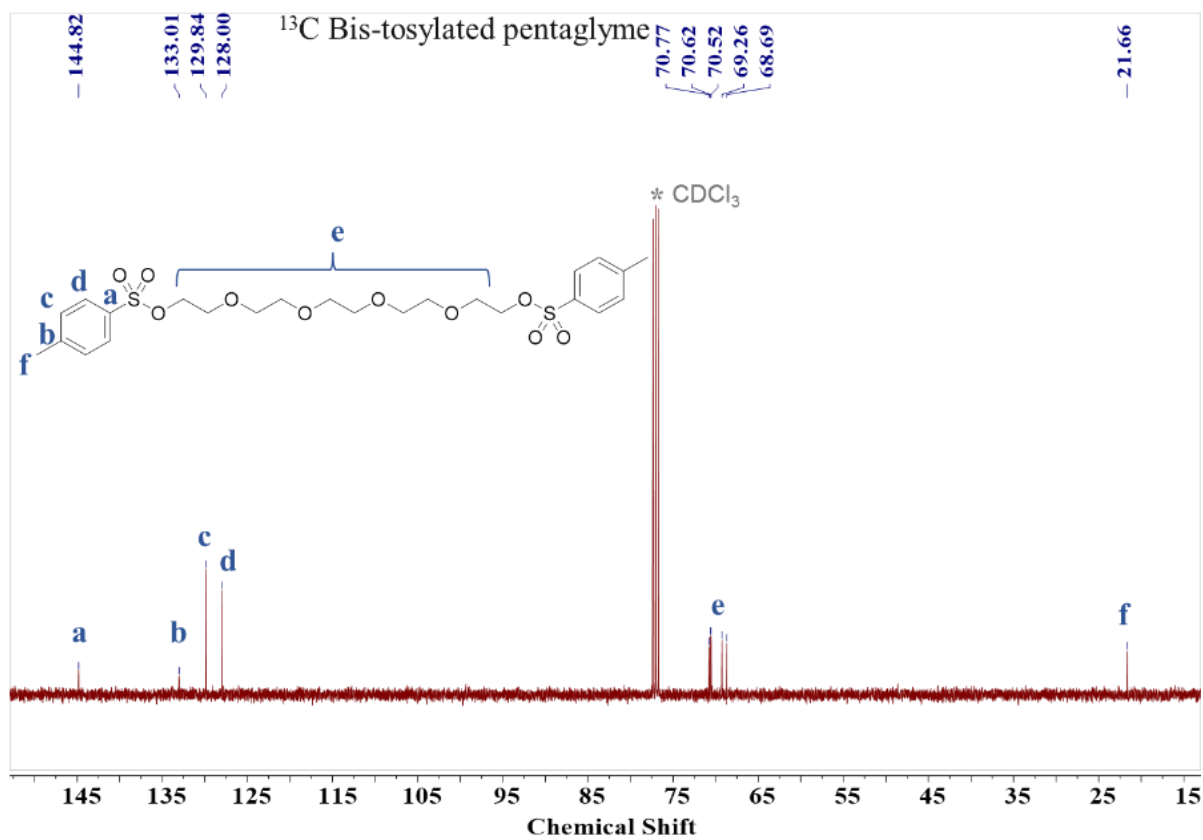
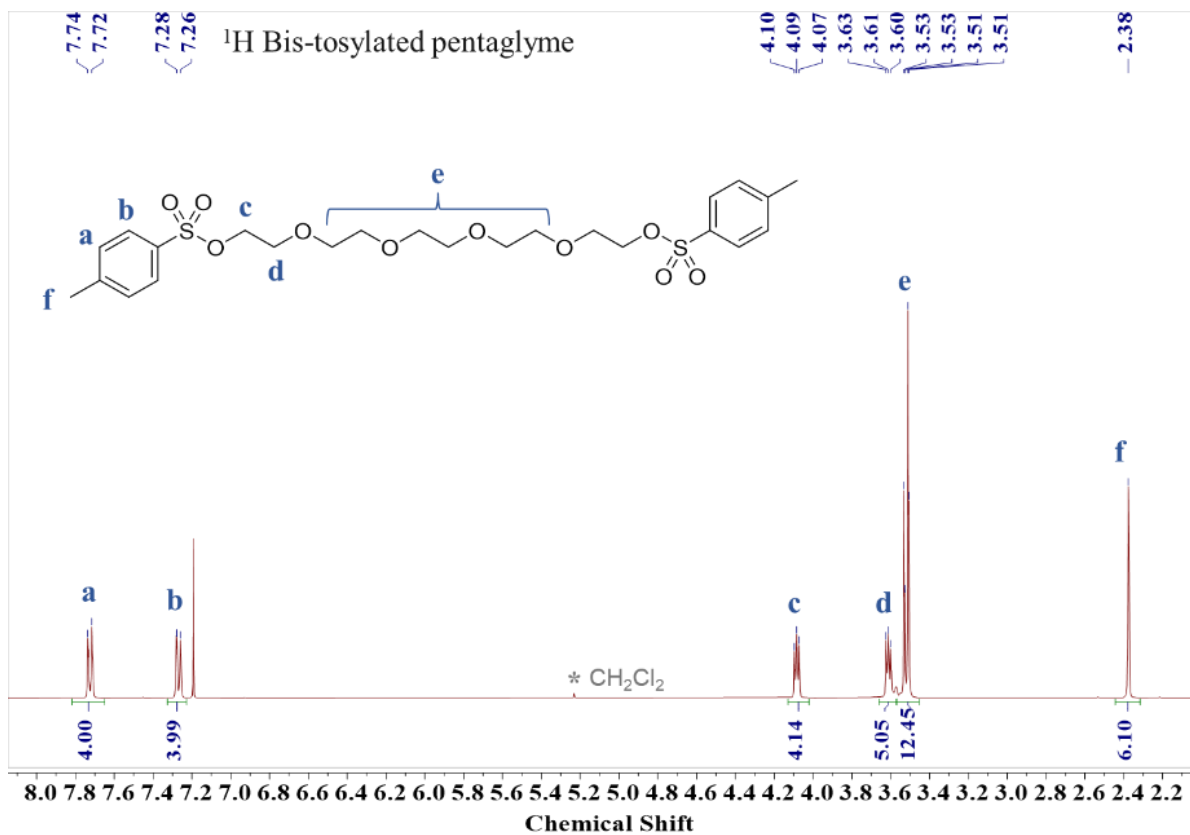


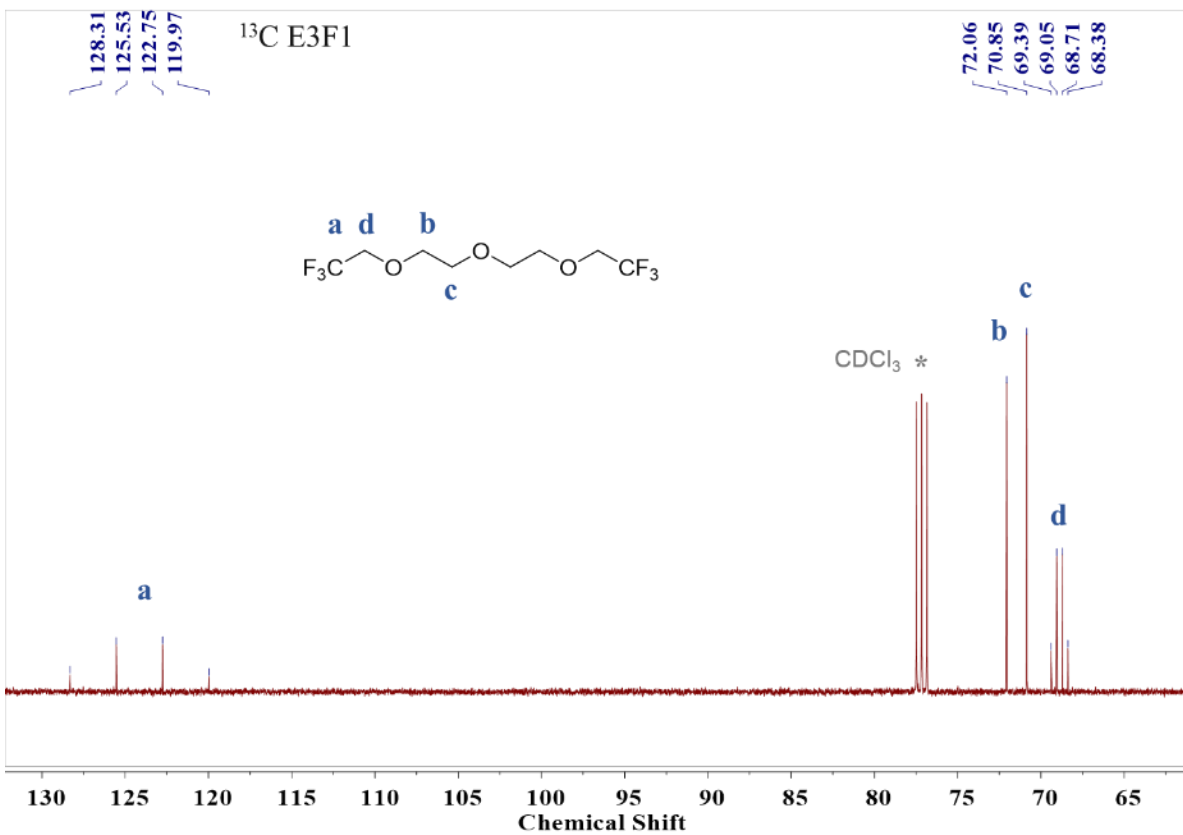
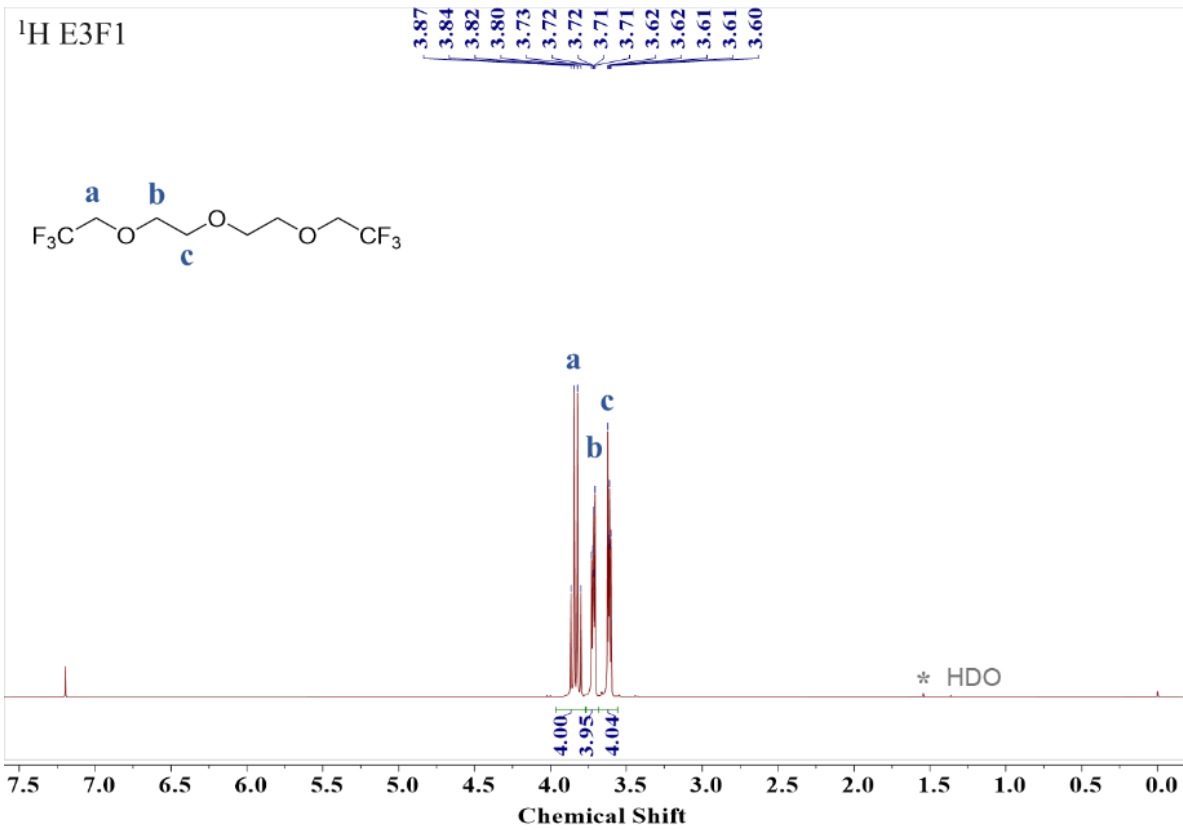
Figure S1. Synthetic route of F1 and F2 compounds. 1) Glycols were tosylated by tosyl chloride with the presence of excess amount of potassium hydroxide. 2) Fluorinated alcohols were deprotonated and then reacted with tosylated glycols to yield F1 and F2 compounds. Detailed synthetic procedure can be found in the Experimental Section.

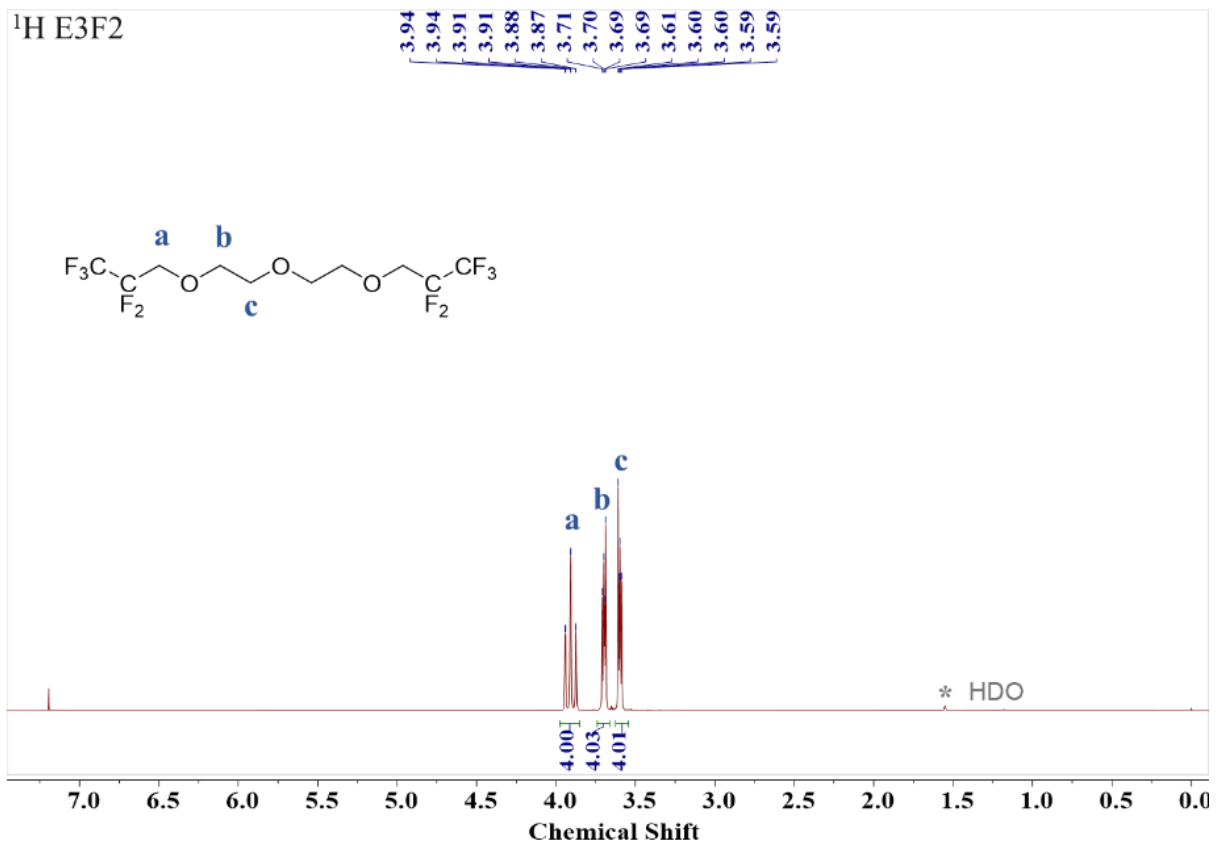
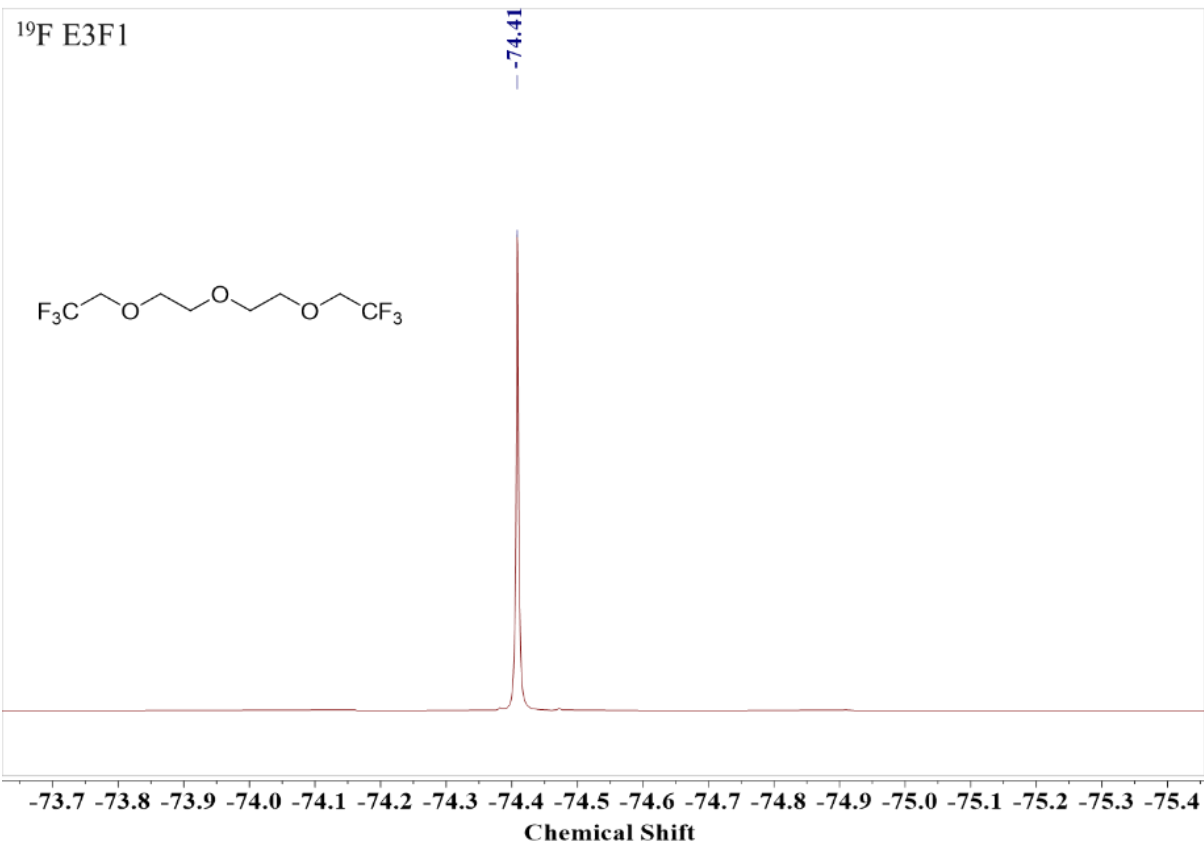


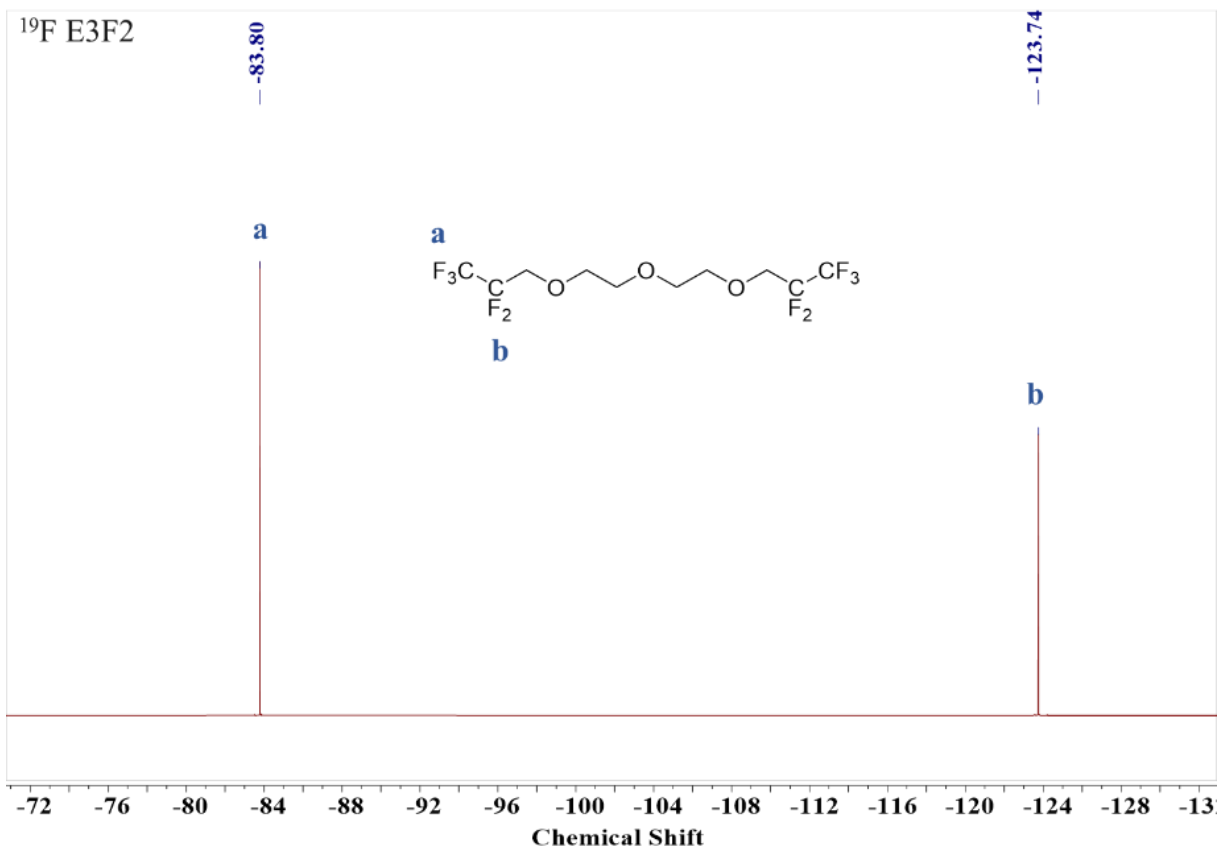
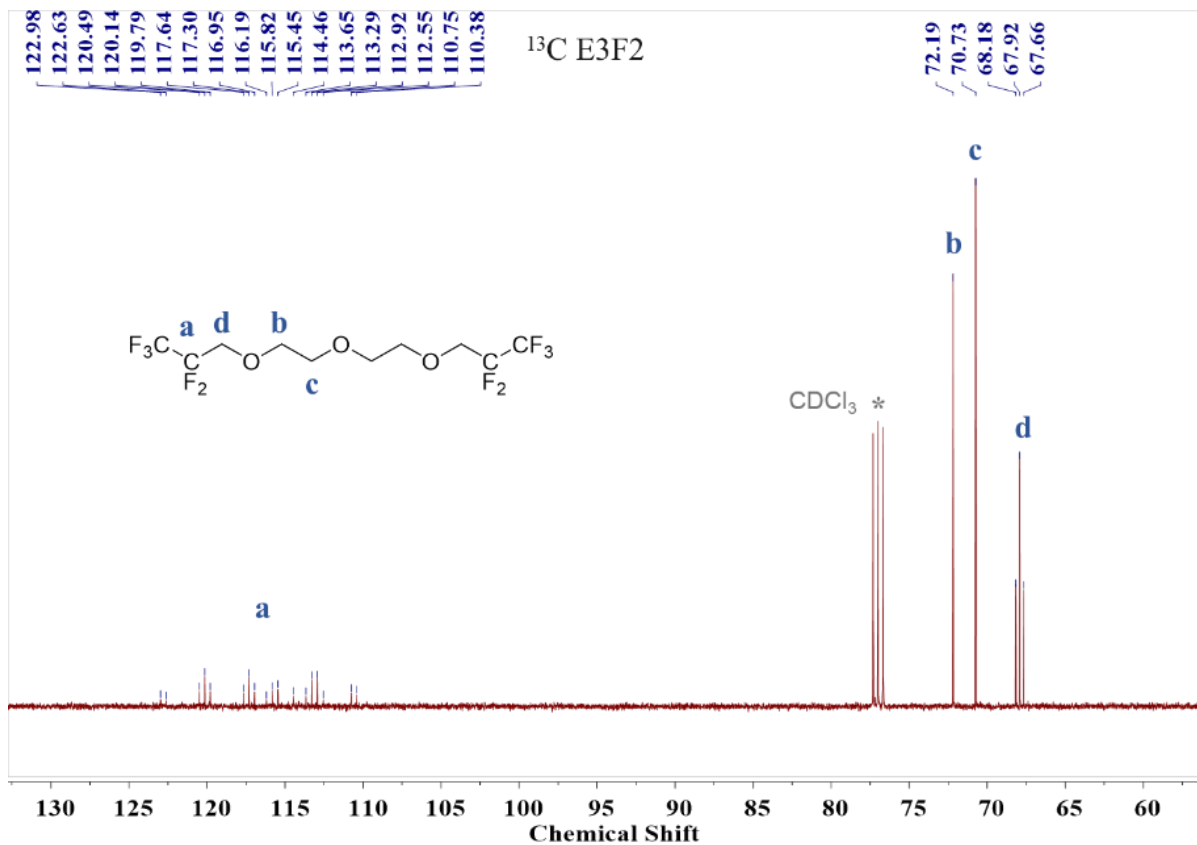


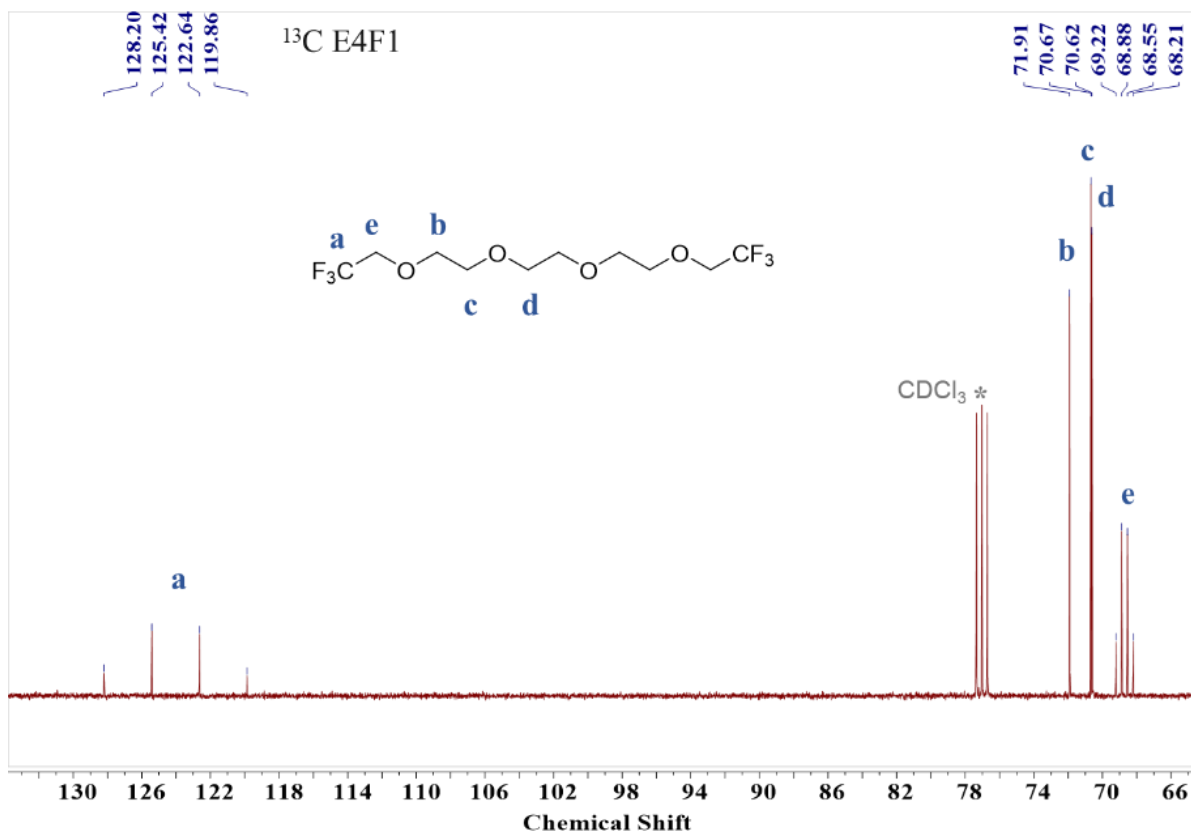
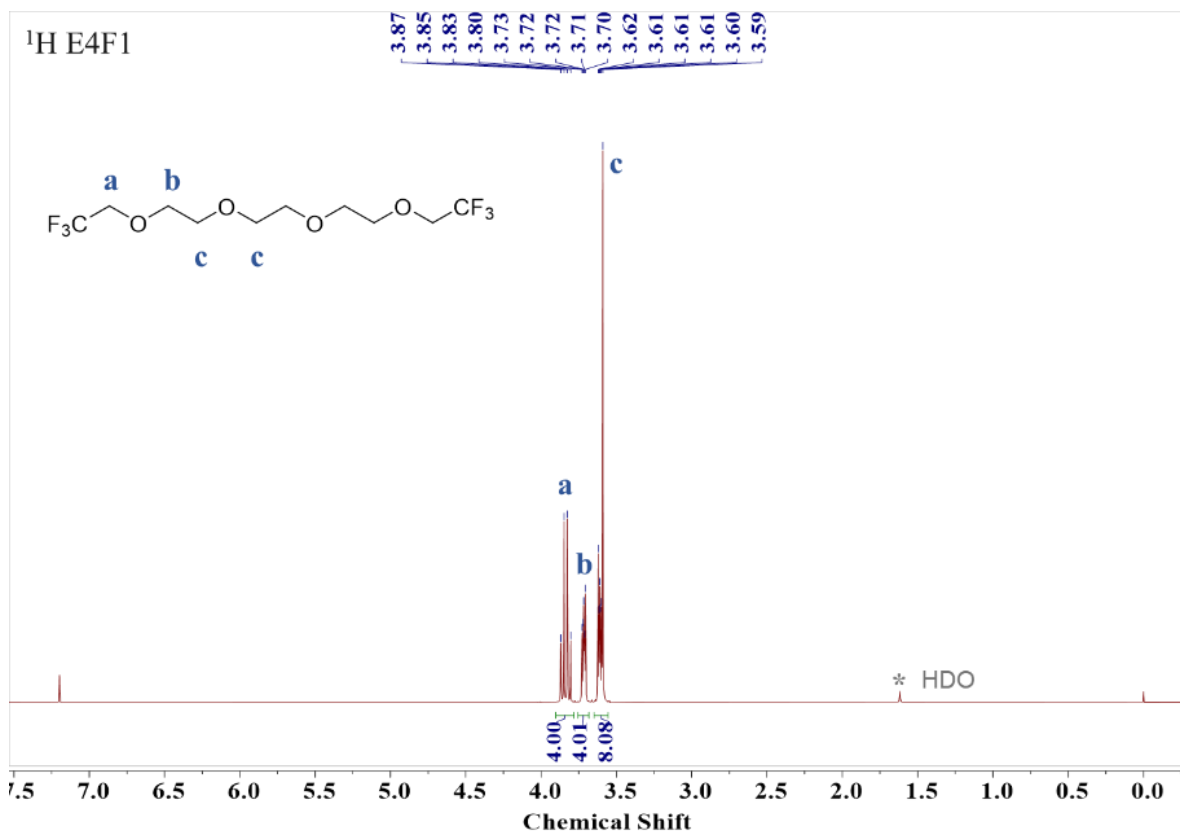


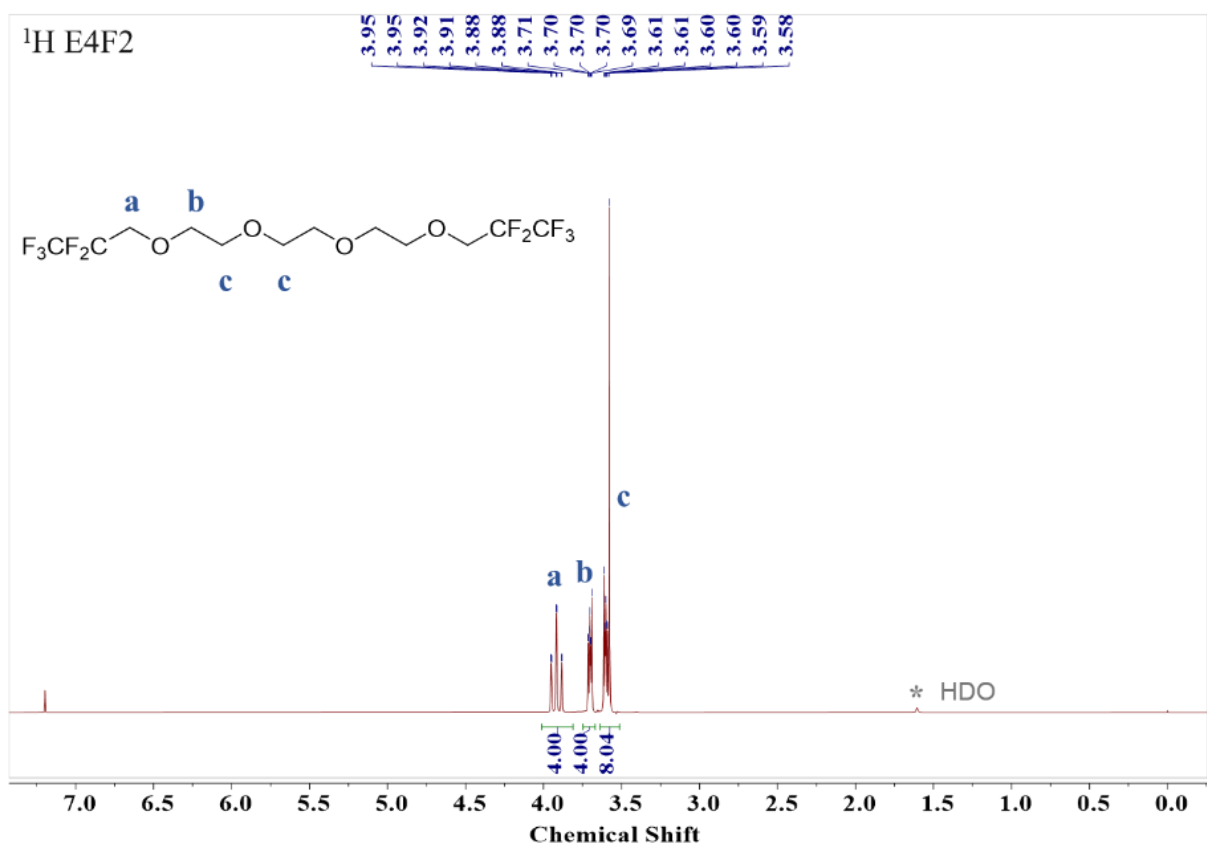
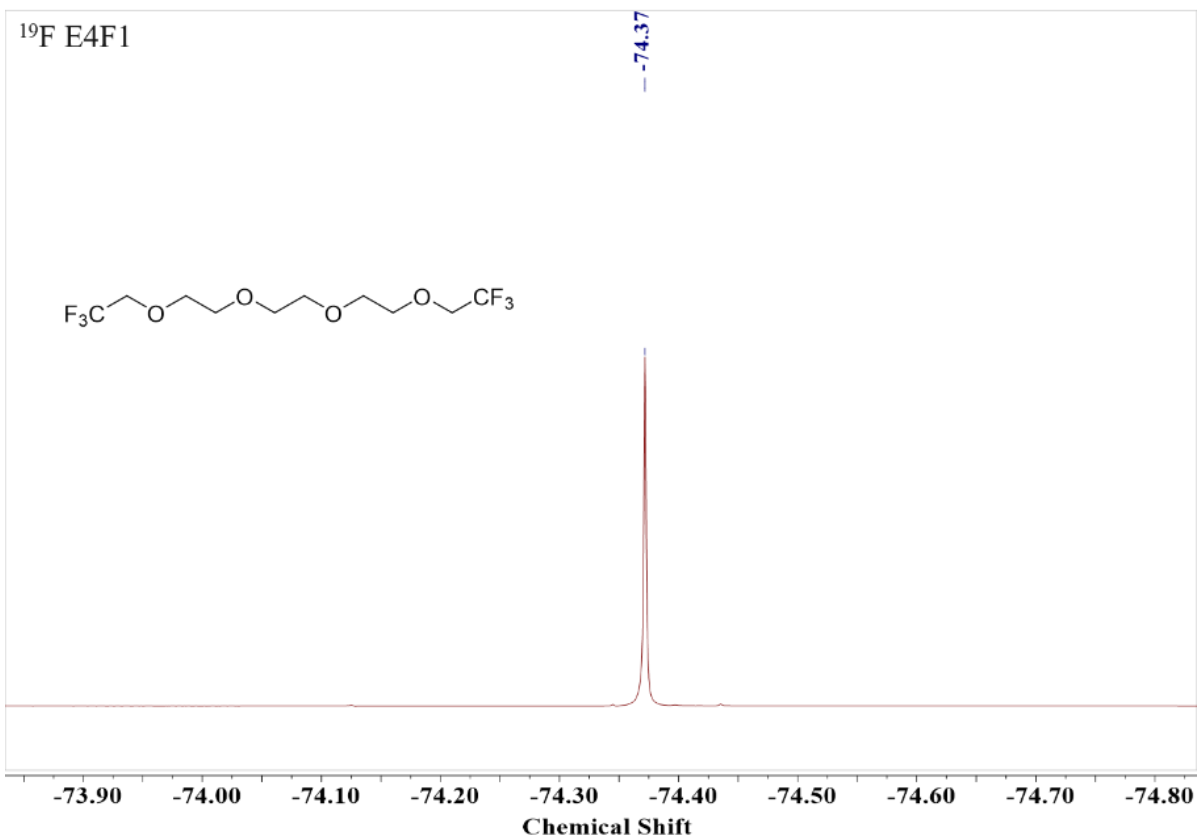


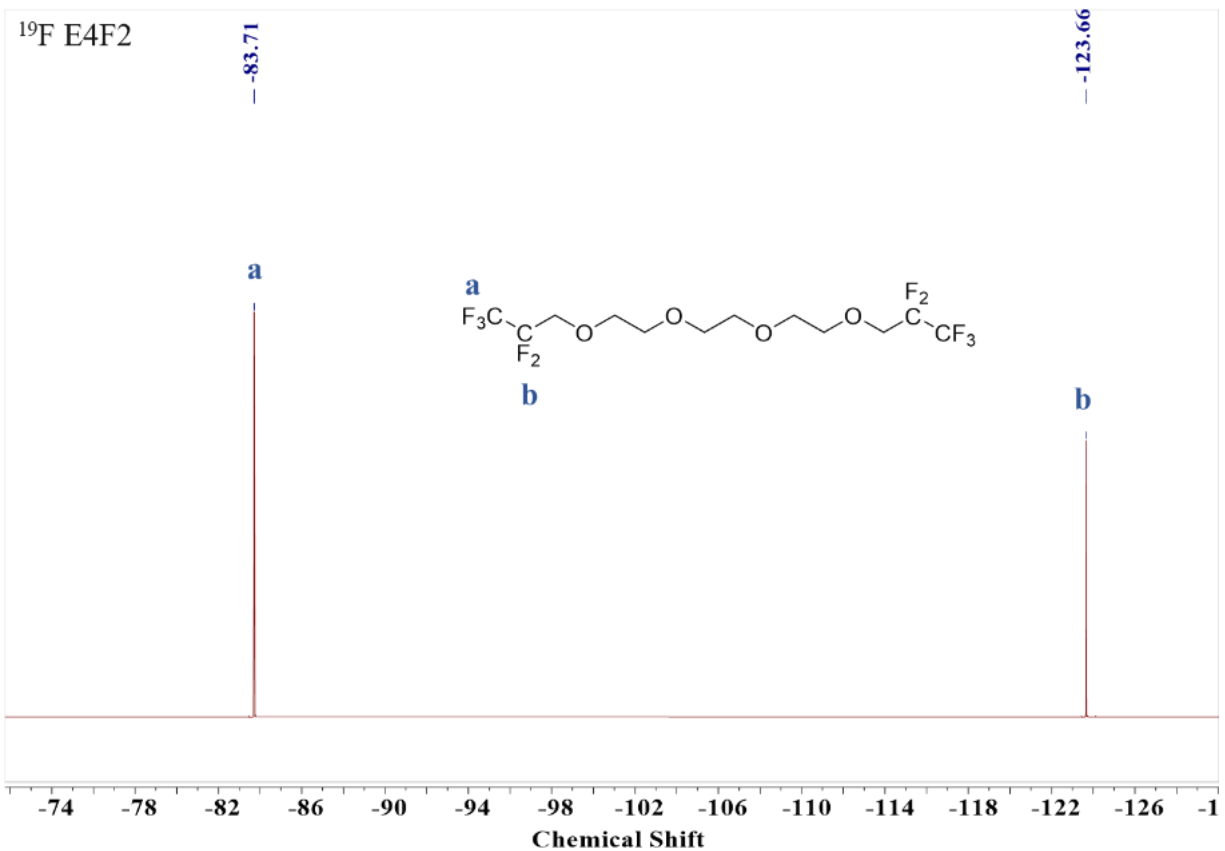
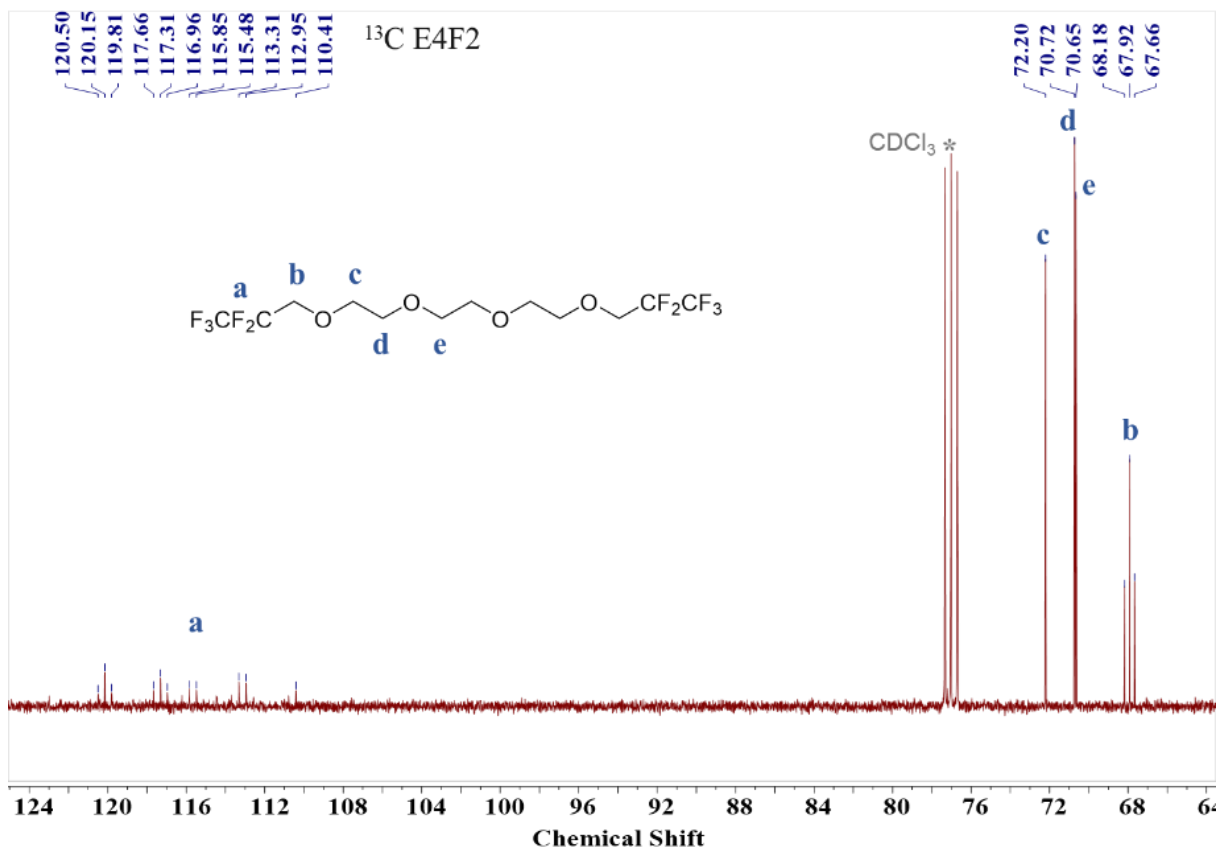


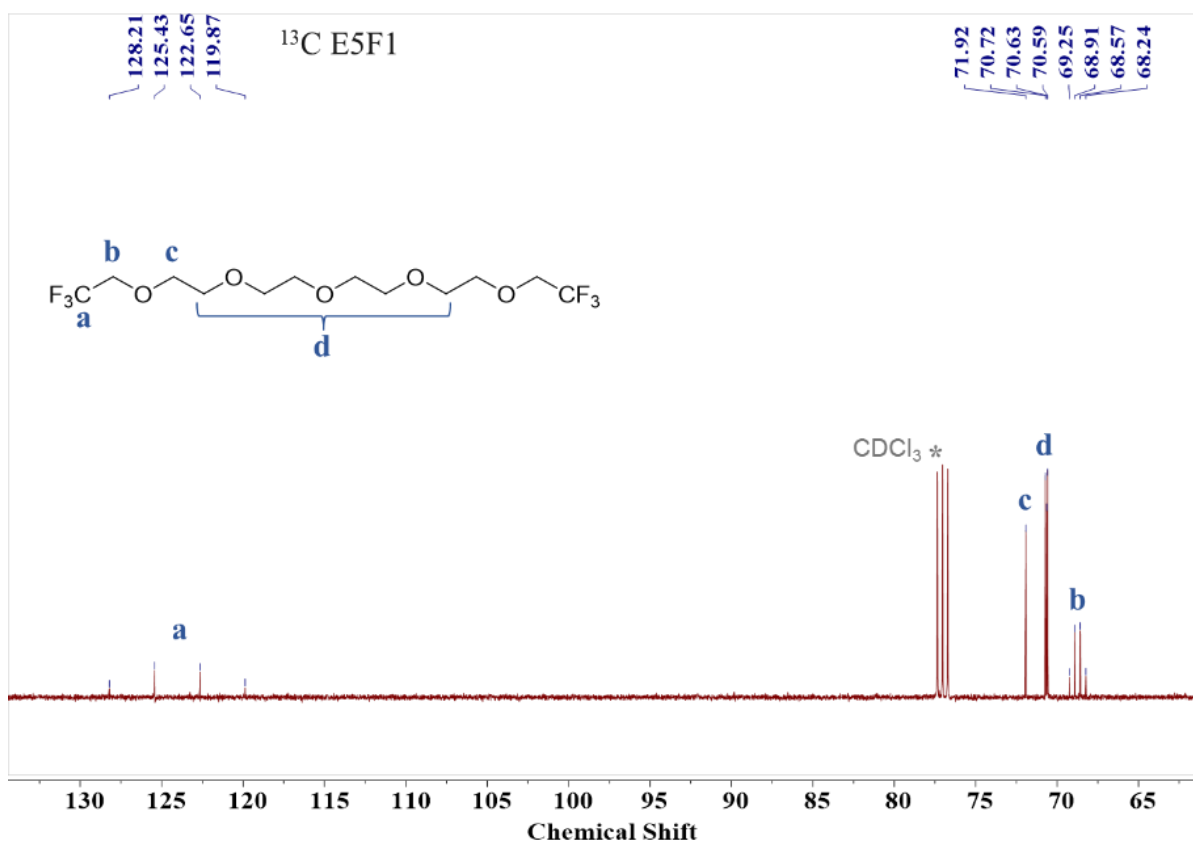
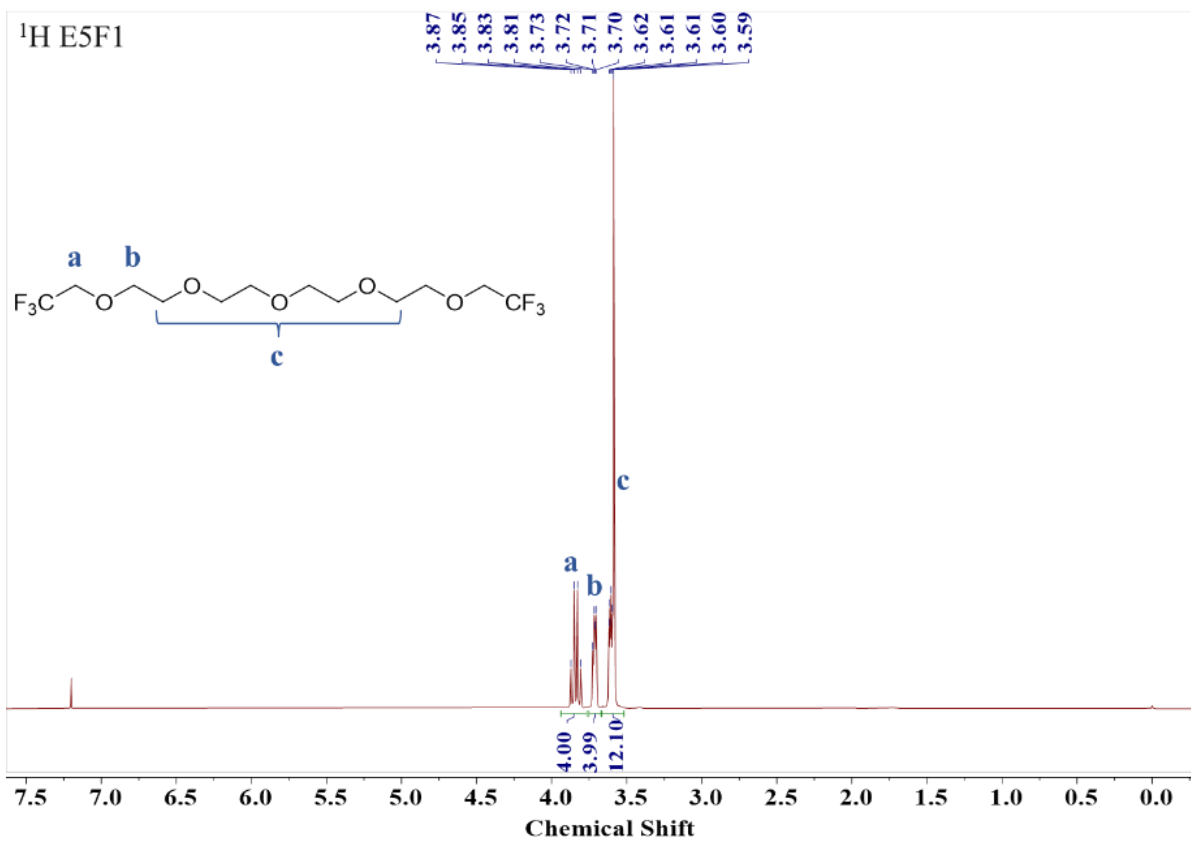


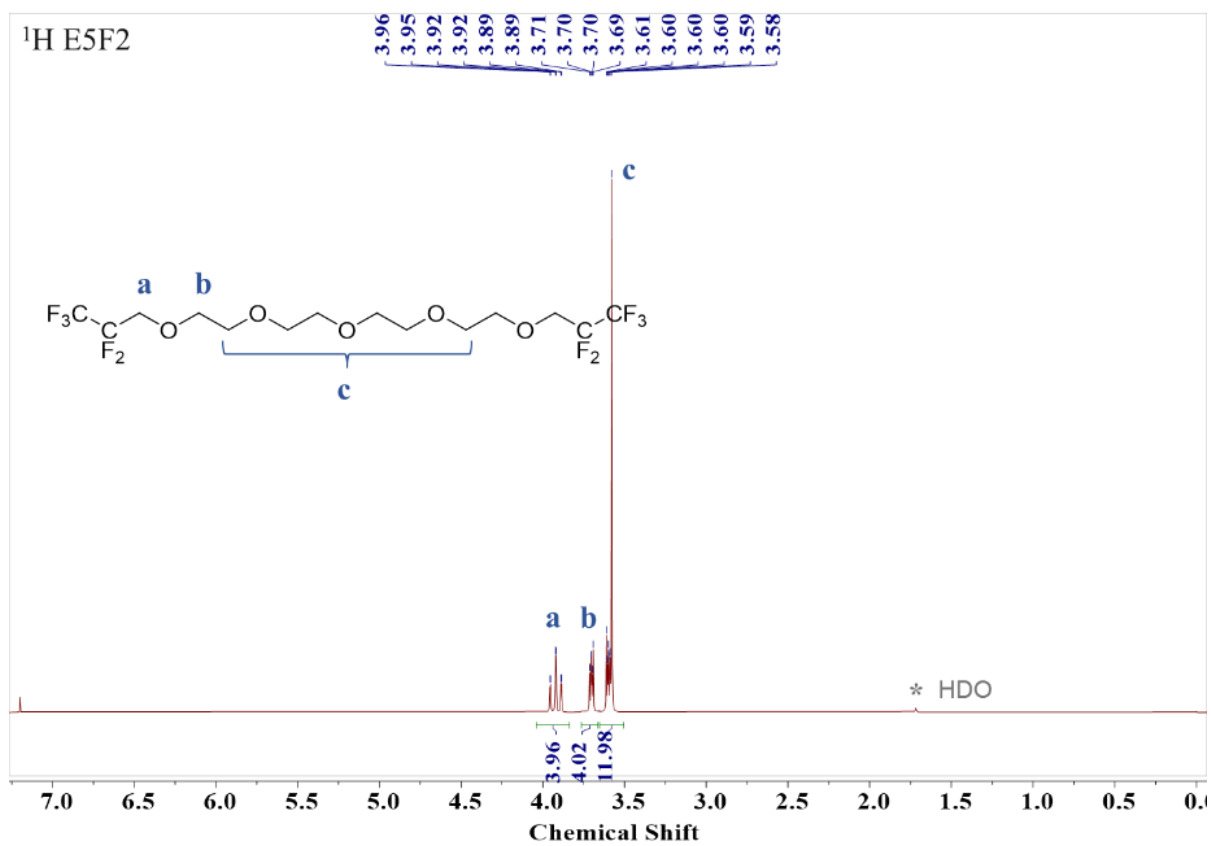
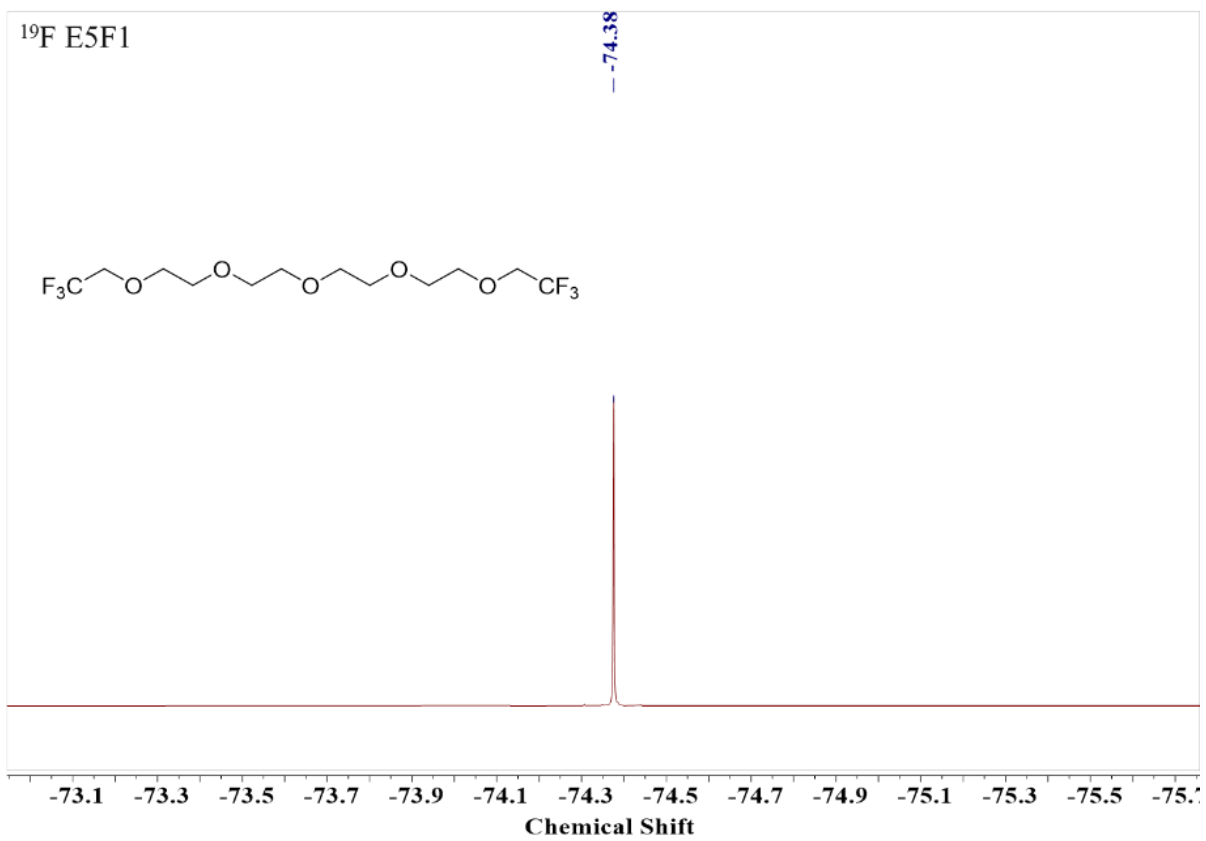


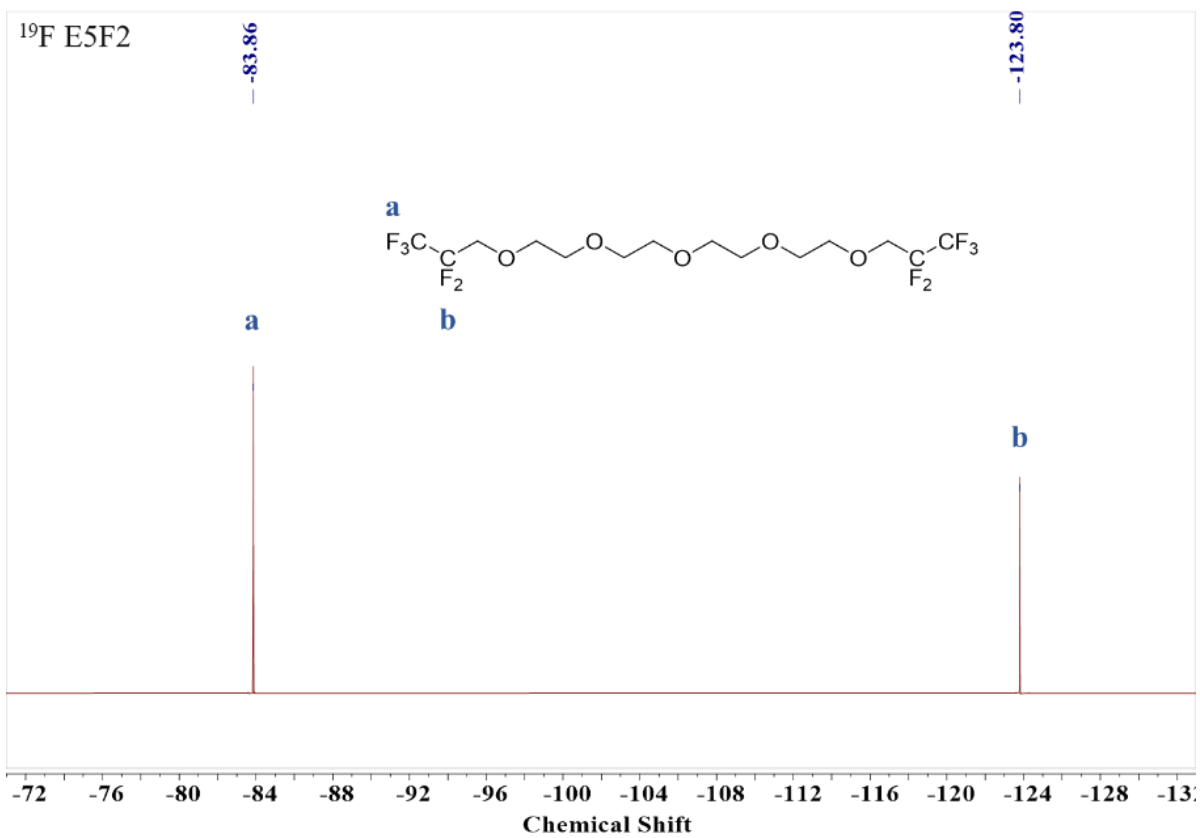
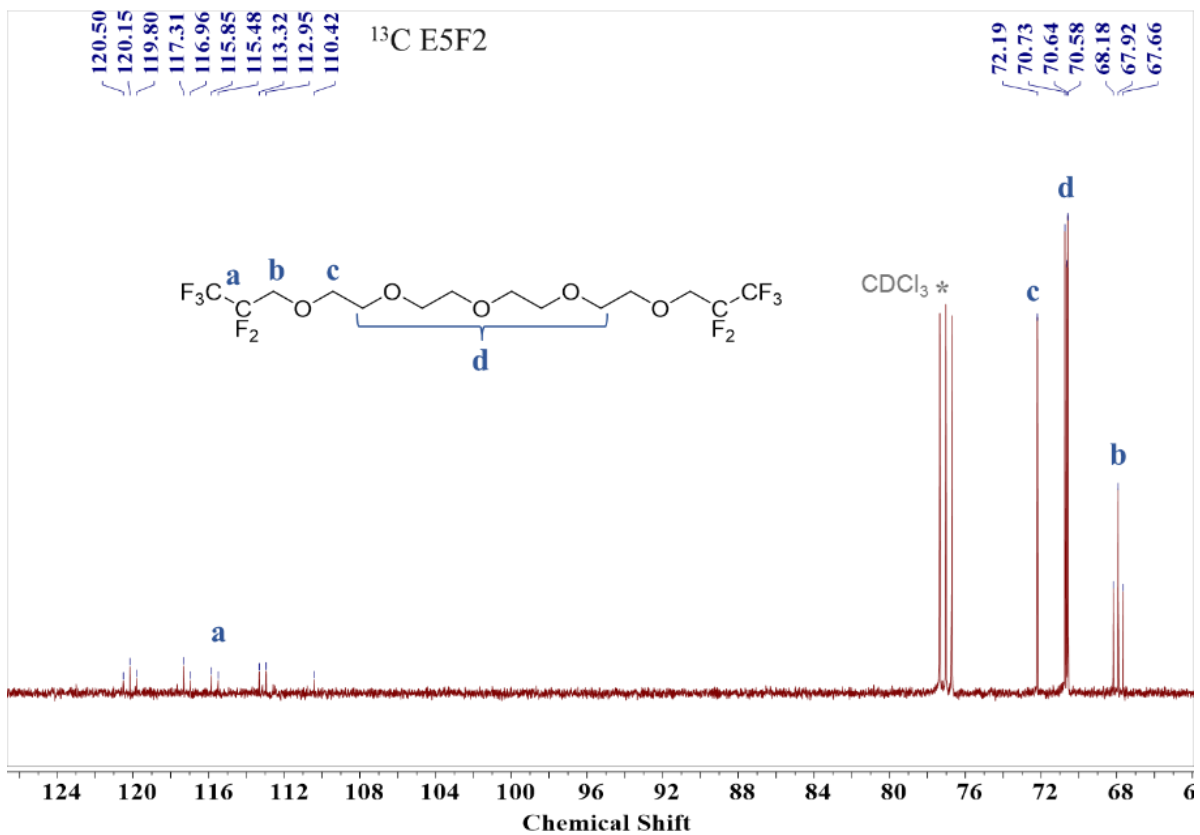


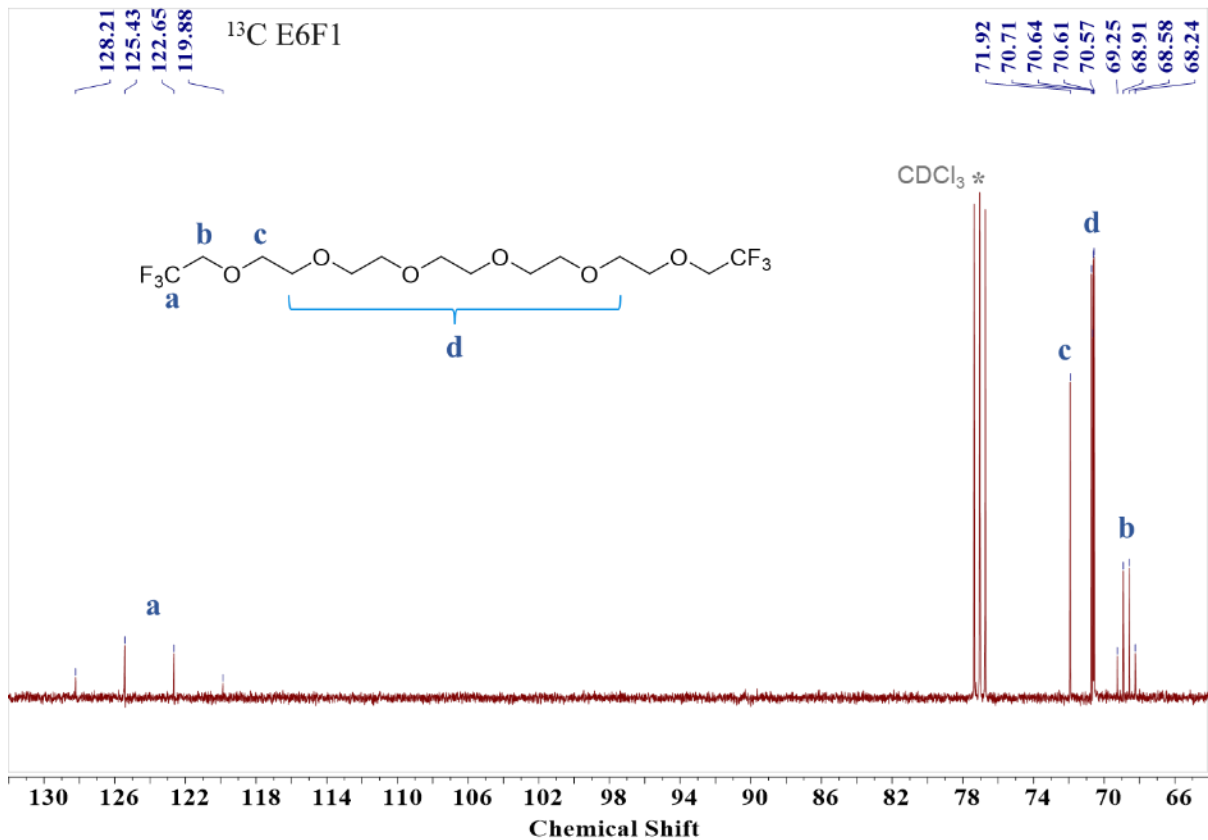
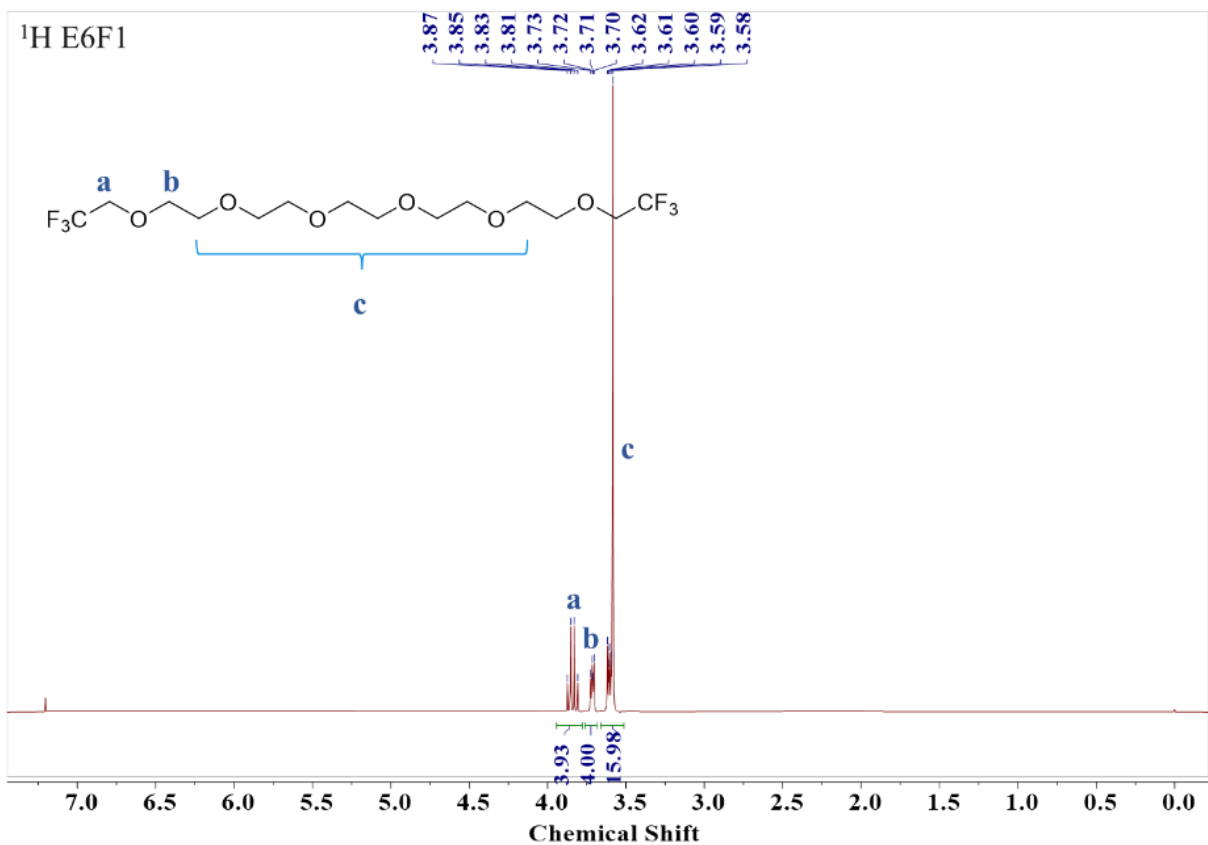












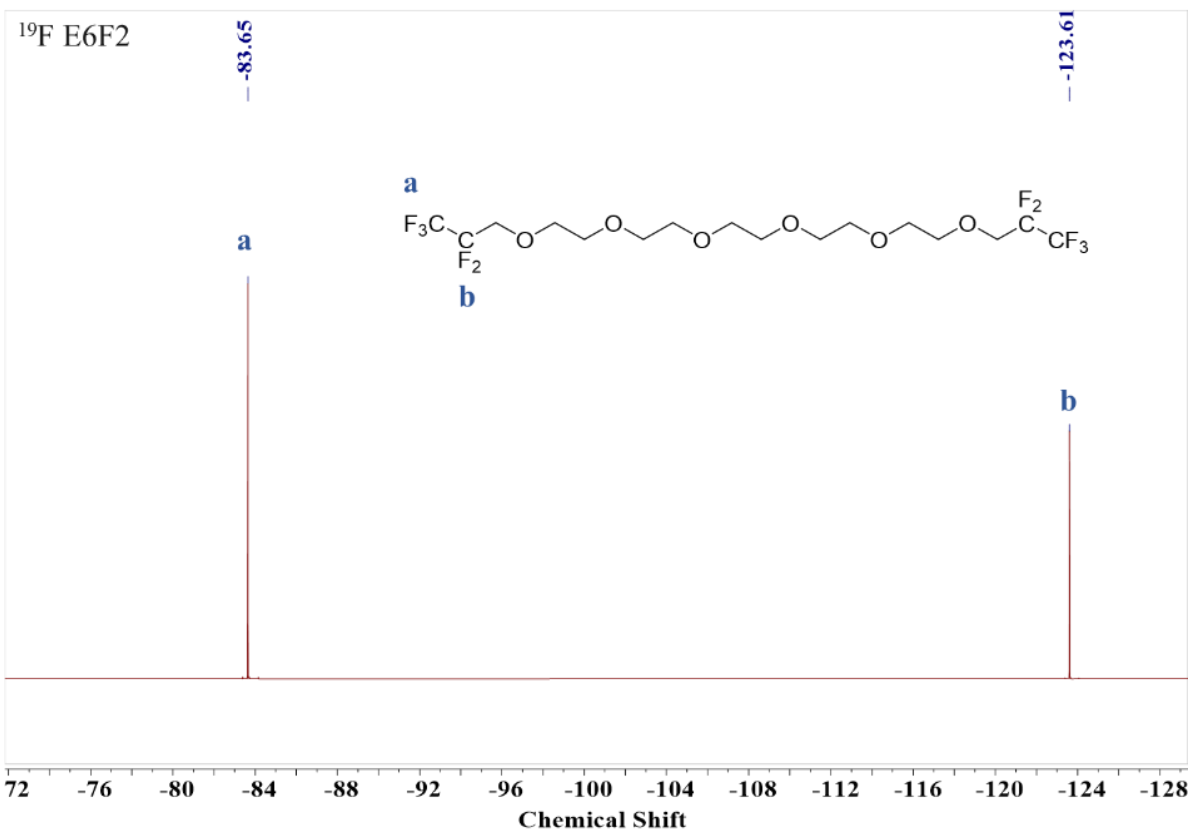
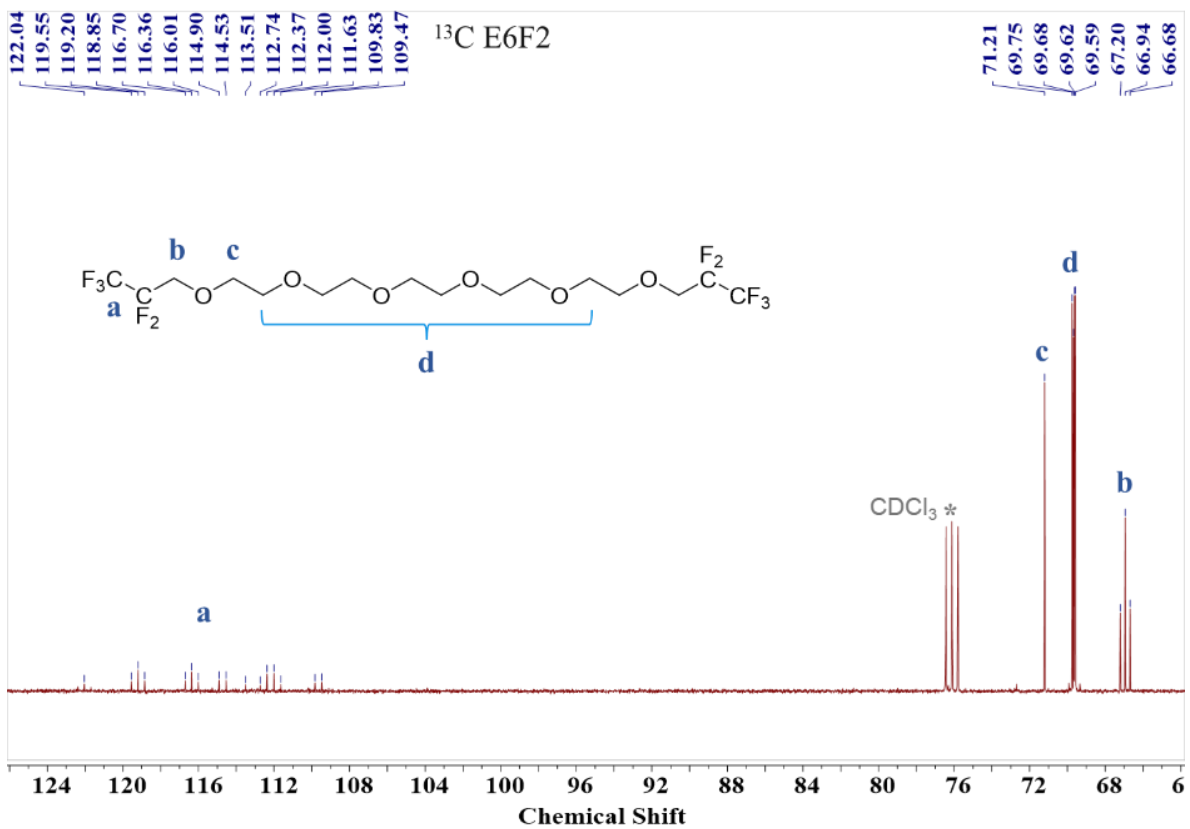
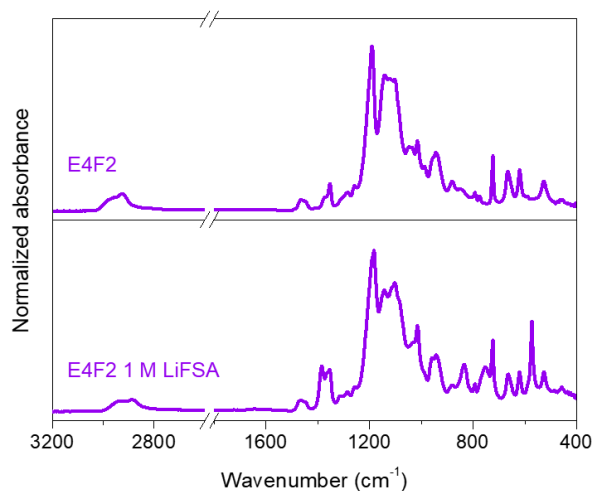
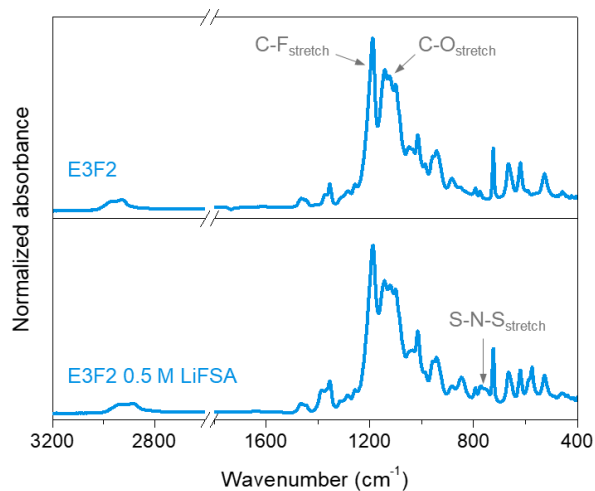
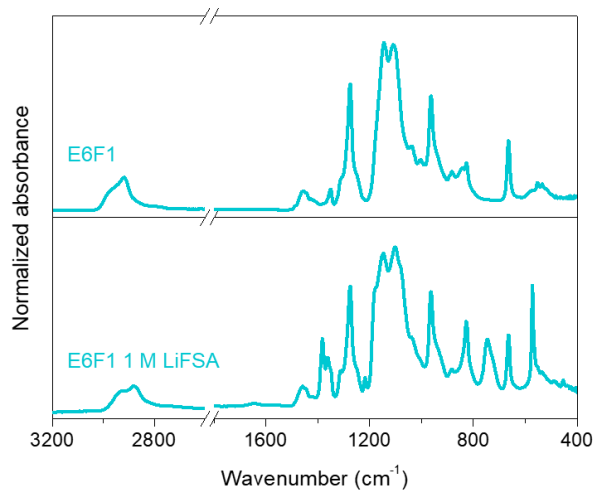
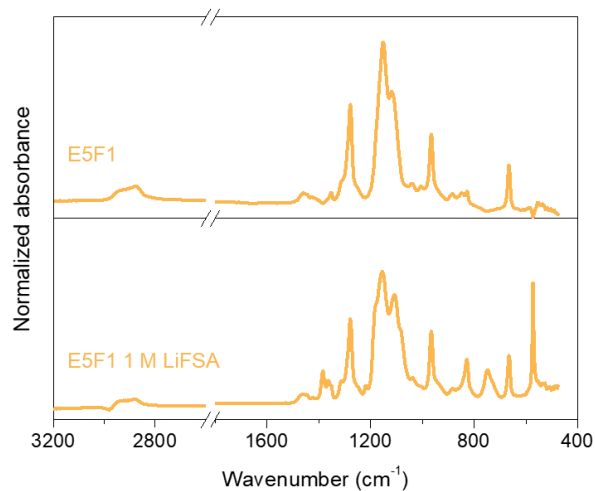
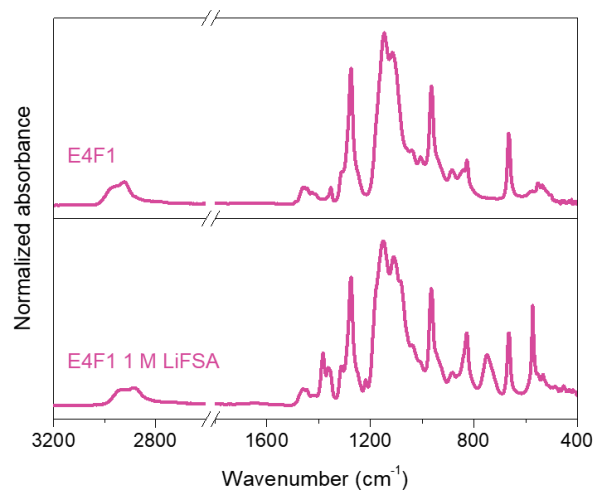
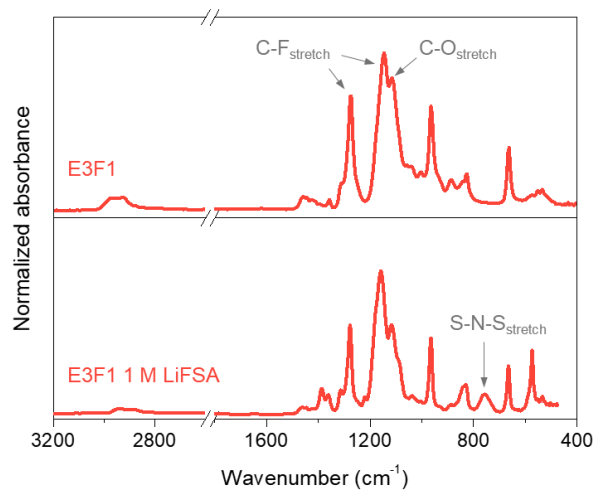


Figure S2. Nuclear magnetic resonance (NMR) spectroscopy of synthesized compounds. ^1H , ^{13}C , and ^{19}F (proton decoupled) NMR spectra were taken on a Bruker Ascend 9.4 T / 400 MHz instrument. NMR sample was prepared by dissolving several milligrams of product into 0.5 mL deuterated chloroform.



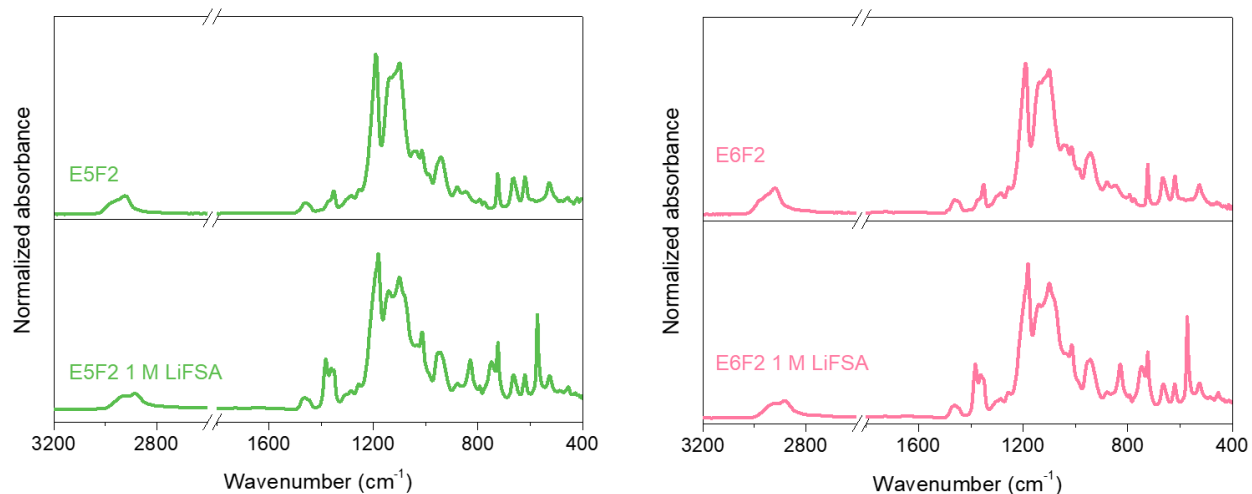
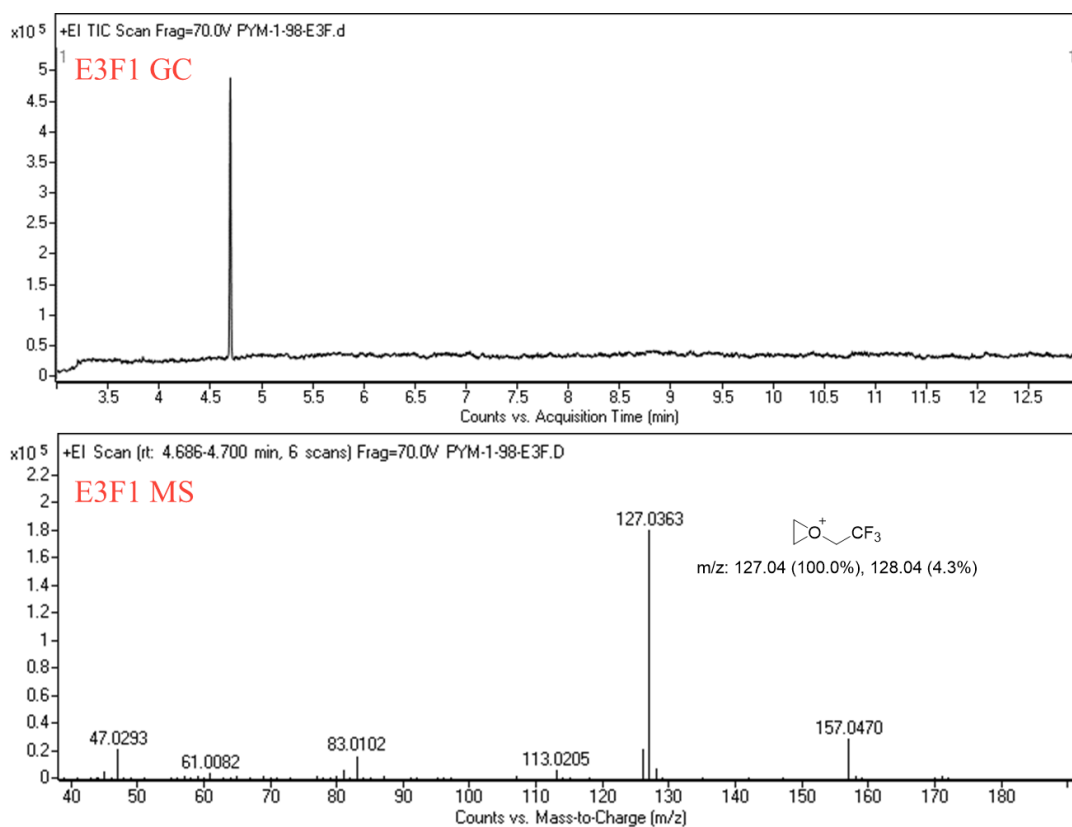
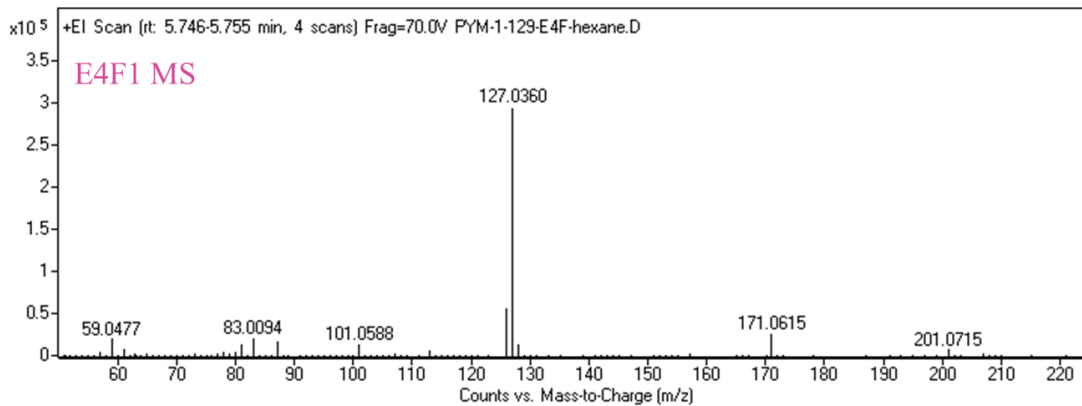
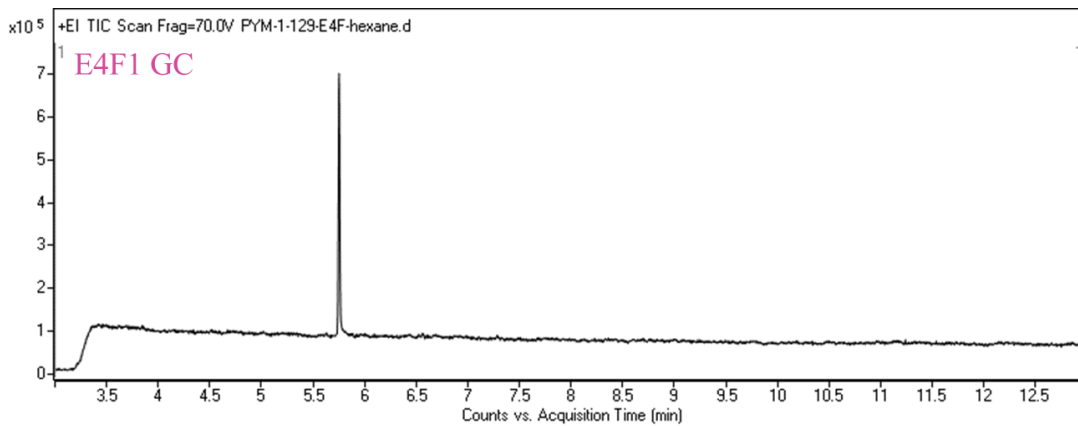
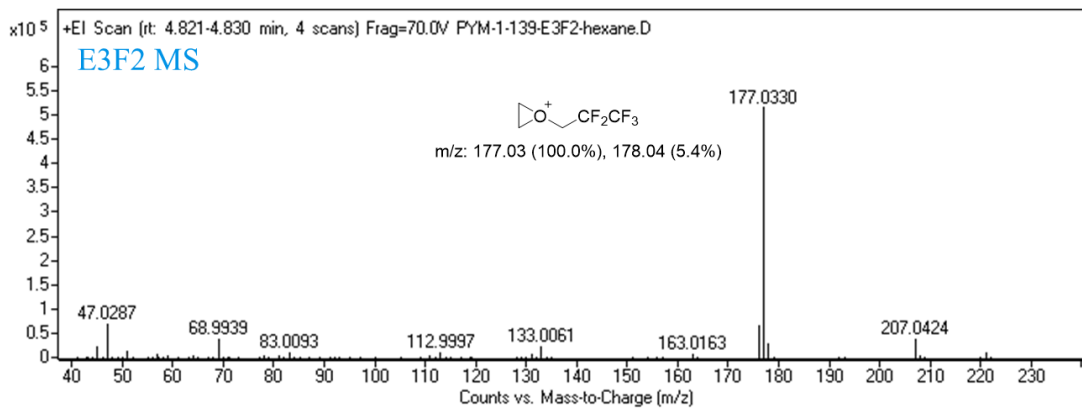
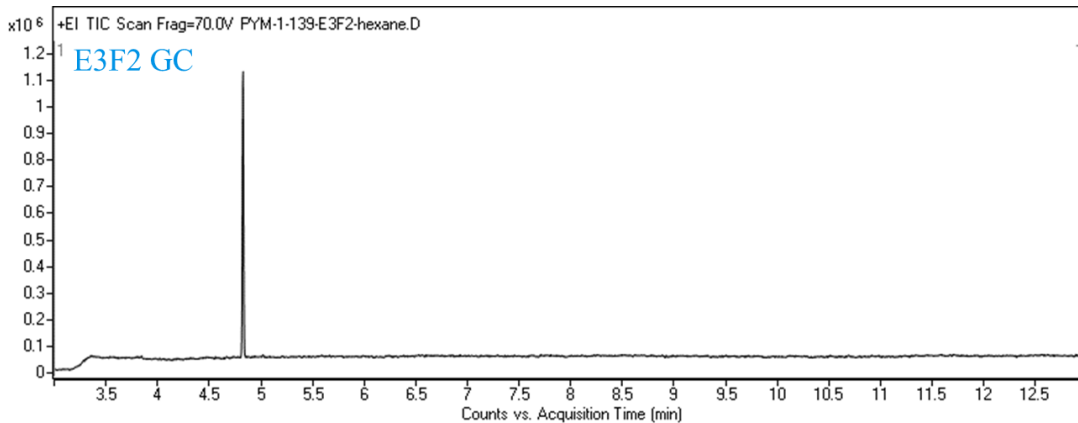
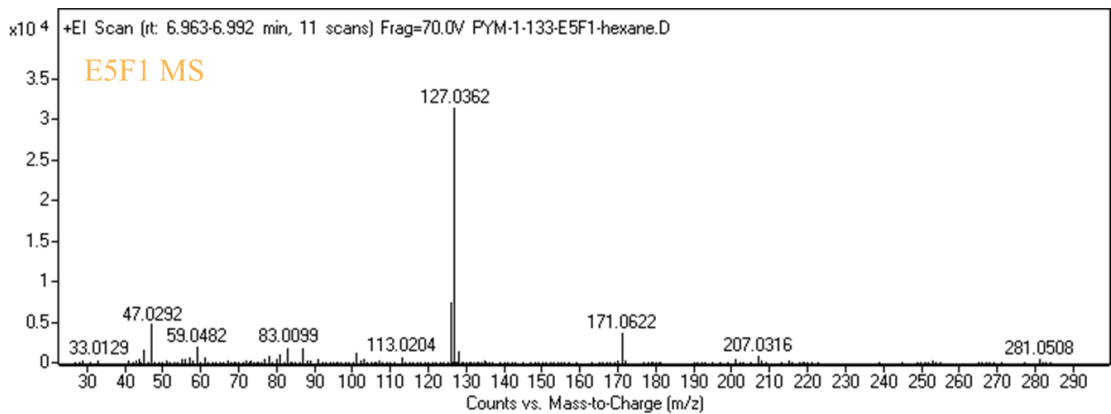
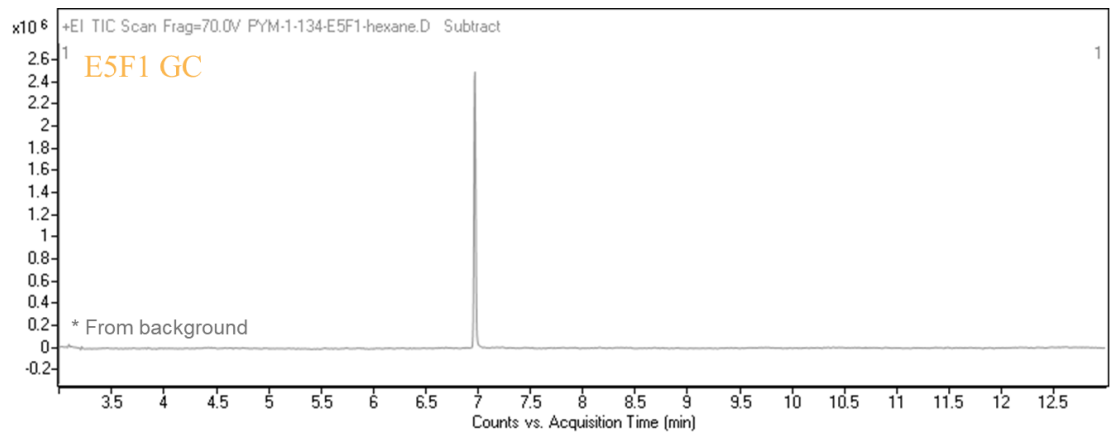
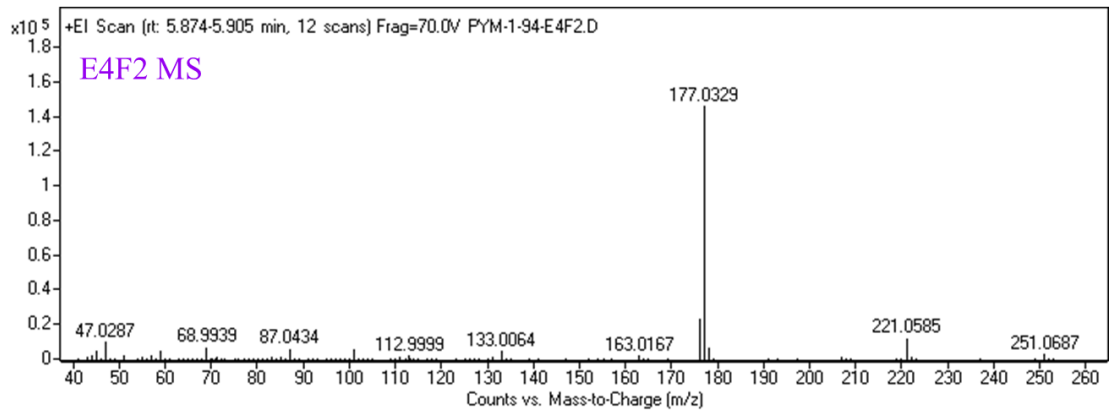
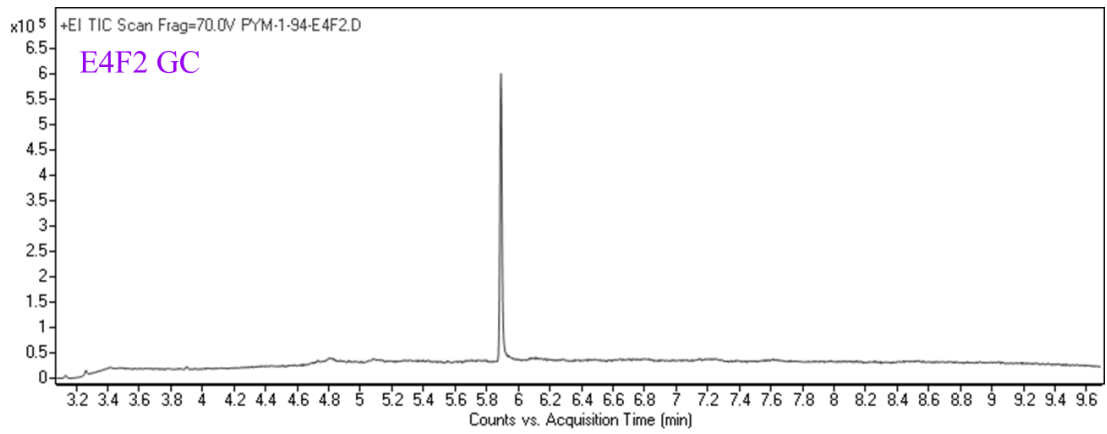
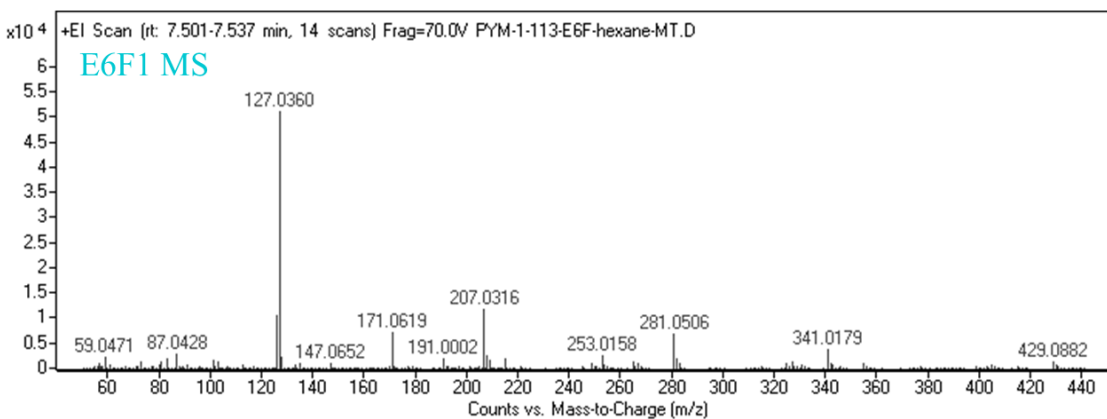
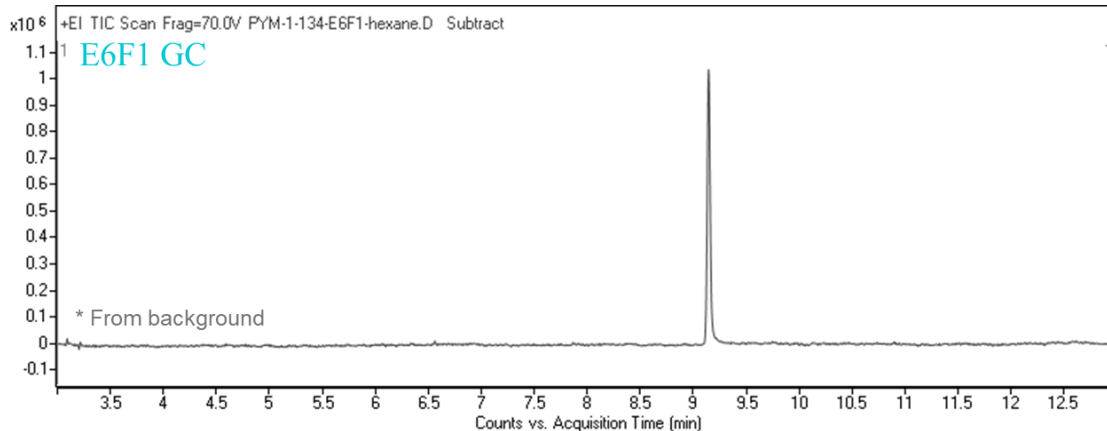
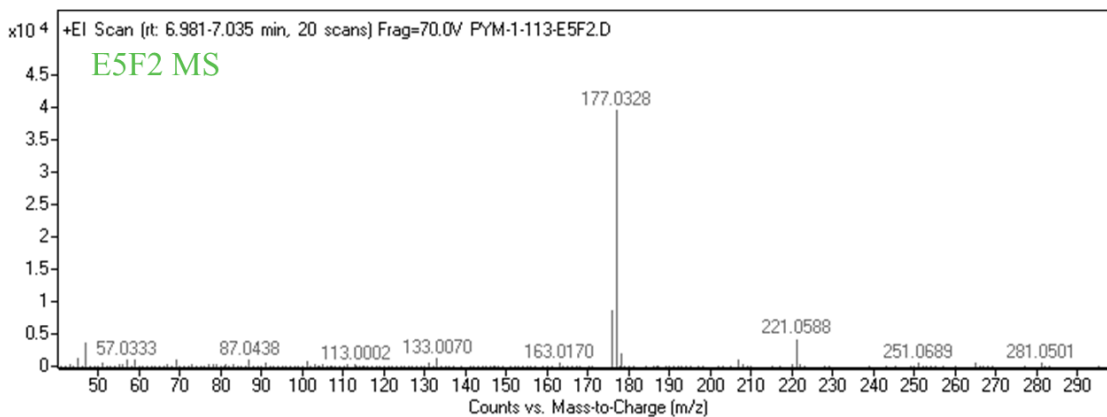
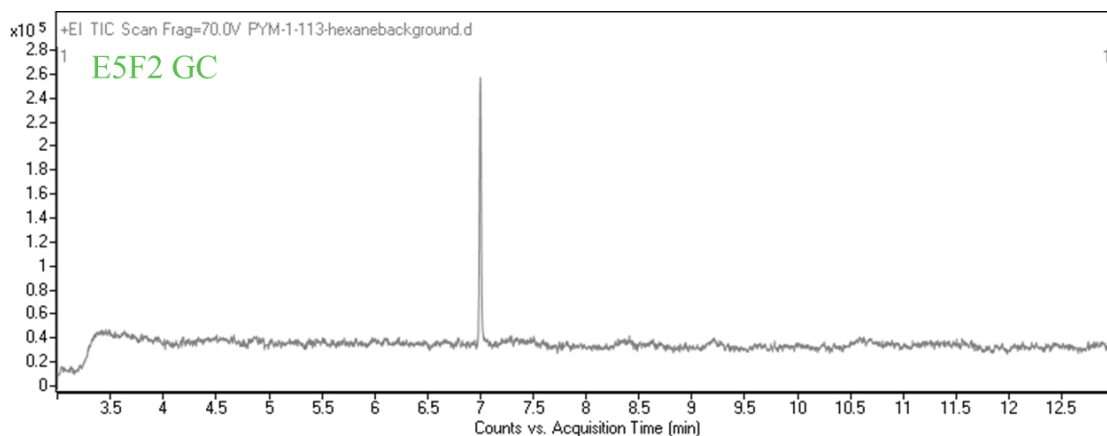


Figure S3. FT-IR spectra of the compounds and their LiFSA solutions. The functional groups indicated in E3F1 correspond to the same functional groups in E4F1, E5F1, and E6F1. The functional groups indicated in E3F2 correspond to the same functional groups in E4F2, E5F2, and E6F2.









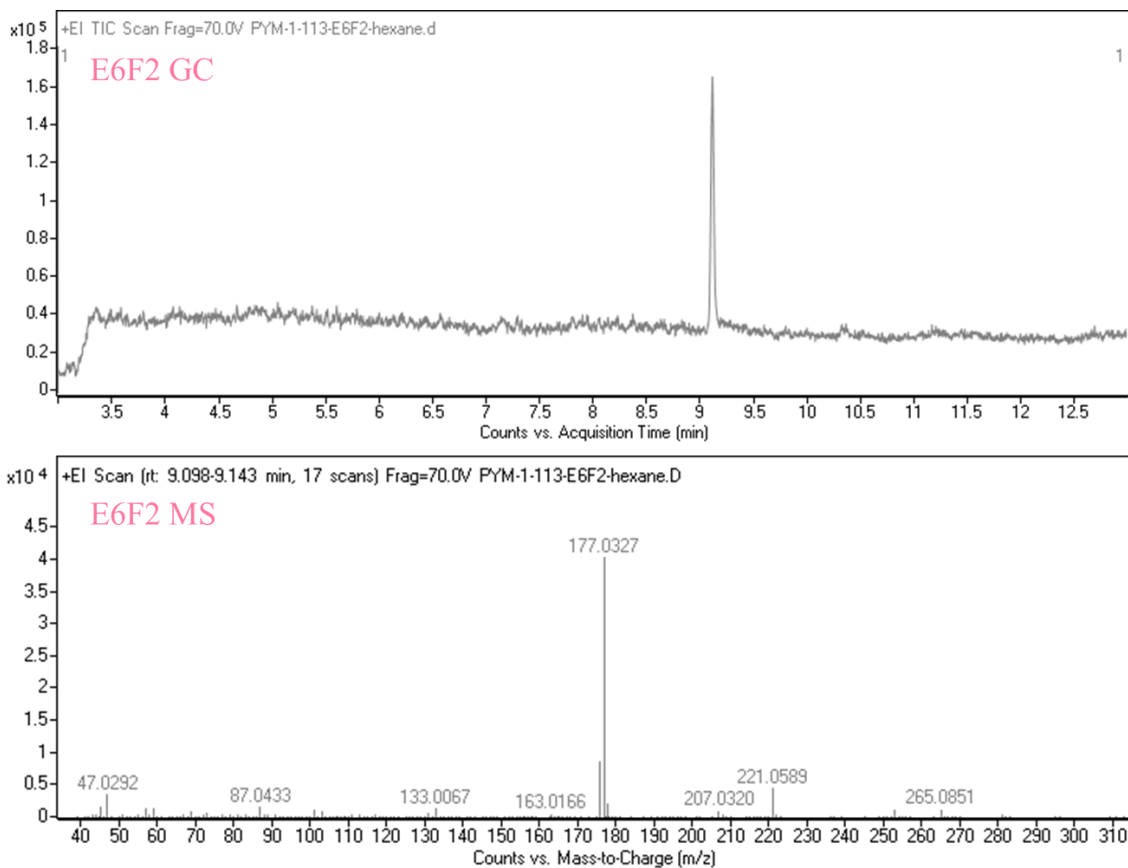


Figure S4. Gas chromatograph-mass spectrometry (GC-MS) data of F1 and F2 compounds. High purity is verified by the absence of impurity peak in GC elution curves. In mass spectra, all F1 compounds have a strong peak at 127.04 g/mol while F2 compounds have a strong peak at 177.03 g/mol. The major fragment is assigned as the corresponding oxonium ions as shown in the mass spectra of E3F1 and E3F2.

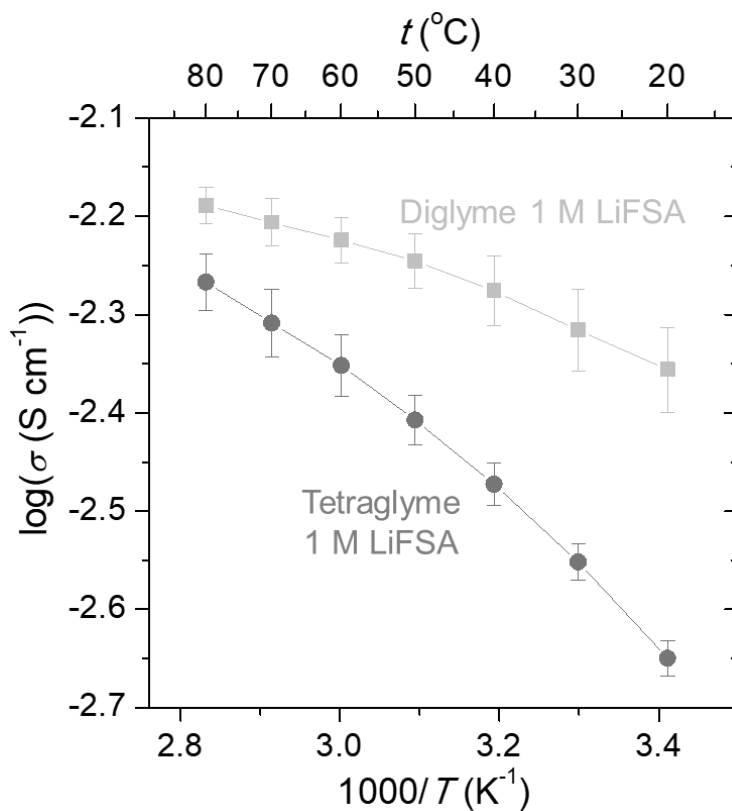


Figure S5. Ionic conductivity as a function of temperature of 1 M LiFSA dissolved in diglyme and tetraglyme. The lines are to guide the eyes. Error bars represent standard deviation from the average of at least 3 different cells. The conductivity of 1 M LiFSA in tetraglyme is close to values reported in literatures.^{1,2}

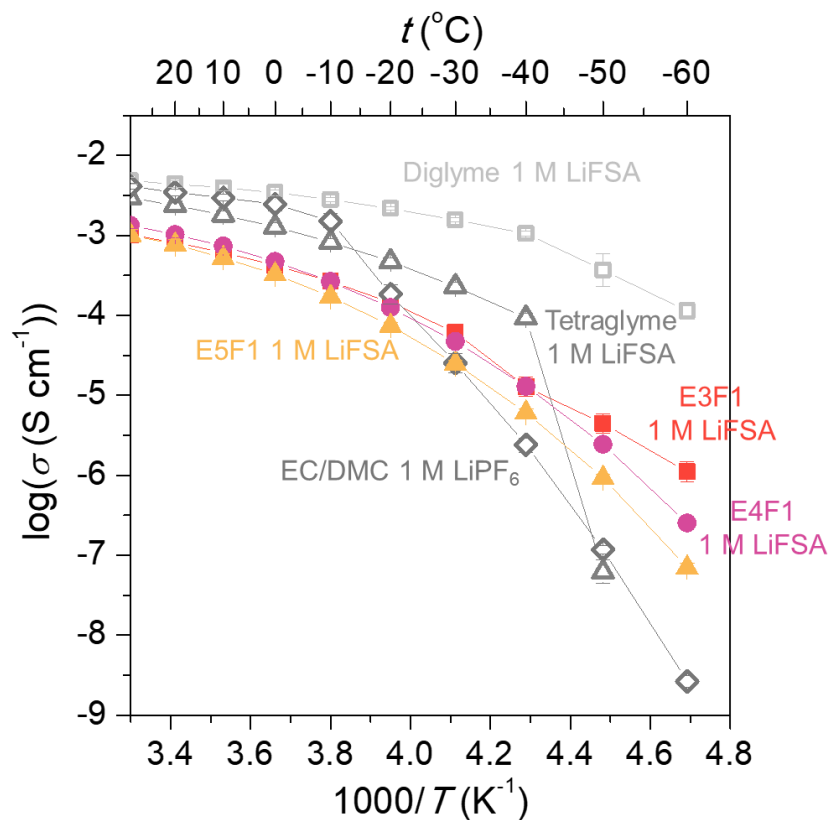


Figure S6. Ionic conductivity in low temperature range. The conductivity was measured by electrochemical impedance spectroscopy (EIS) following procedure described in experimental section. For the 1 M LiPF₆ in EC/DMC (50:50 v/v) electrolyte, a Whatman separator was used for better wetting. Hence a different cell constant of 1 was used.

Because of crystallization (see DSC data in Figure S12), the conductivity of EC/DMC and tetraglyme electrolytes has sharp turns at -10°C and -40°C , respectively. After this transition, their conductivities drop much faster than the fluoroether electrolytes. At -60°C , E3F1 maintains conductivity above 10^{-3} mS cm⁻¹, which is two orders of magnitude higher than that of carbonate electrolyte.

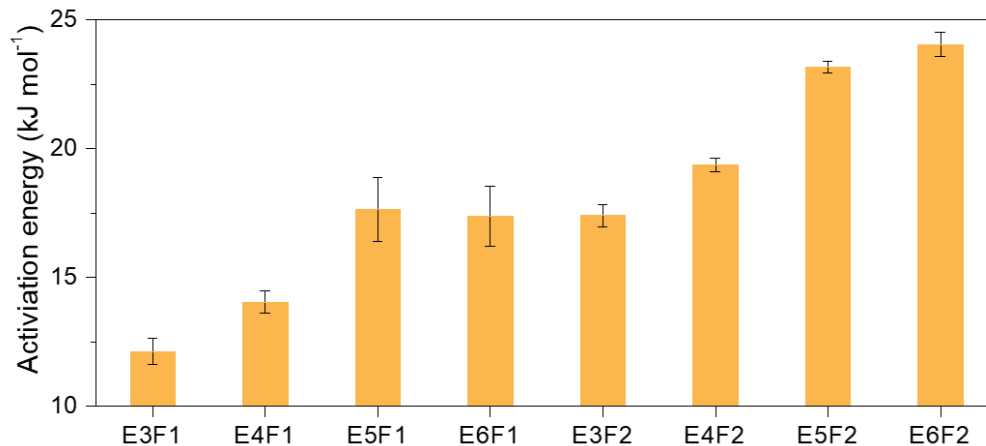


Figure S7. Activation energy of 1 M LiFSA electrolytes. The activation energy of ion transport is extracted from linear fitting of conductivity versus temperature plots shown in the main manuscript Figures 2a and 2b.

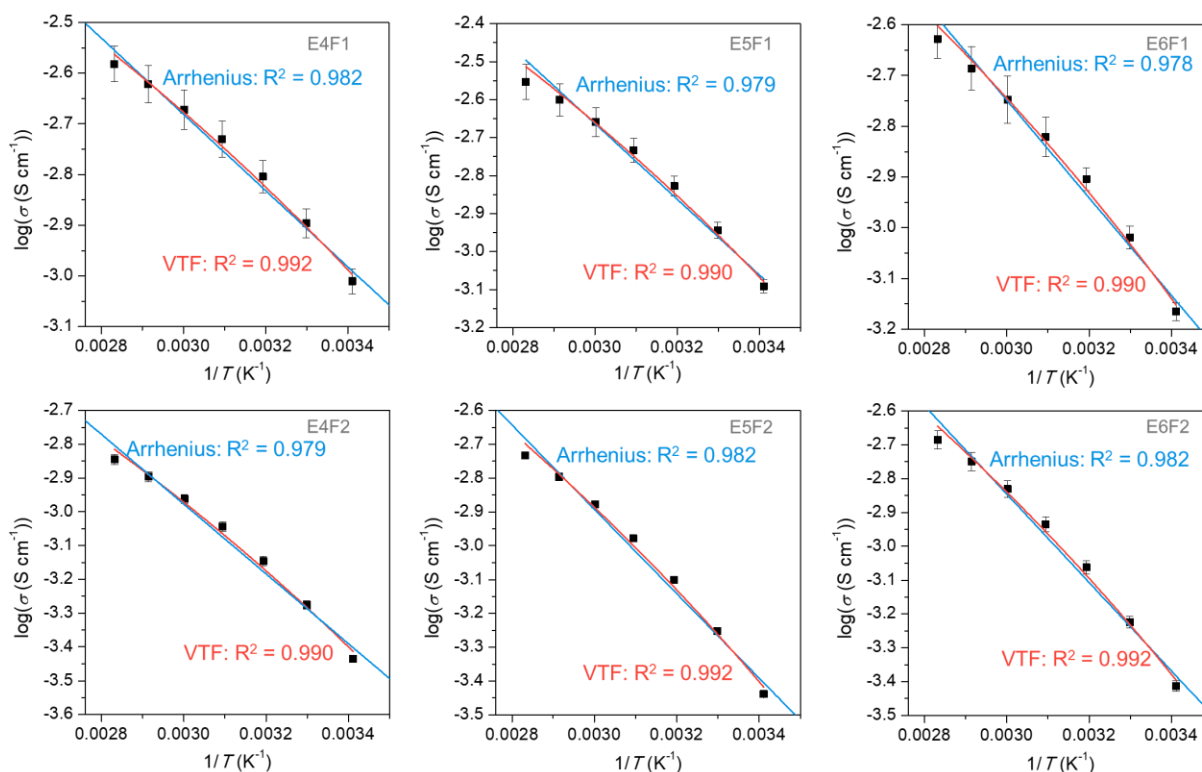


Figure S8. Vogel–Tammann–Fulcher (VTF) and Arrhenius fitting of ionic conductivity data. VTF equation can fit slightly better since it accounts the influence of glass transition temperature (T_g).

Vogel–Tammann–Fulcher equation: $\sigma = Ae^{\frac{-B}{T-T_o}}$, where $T_o = T_g - 50$.

Arrhenius equation: $\sigma = Ae^{-\frac{E_a}{RT}}$.

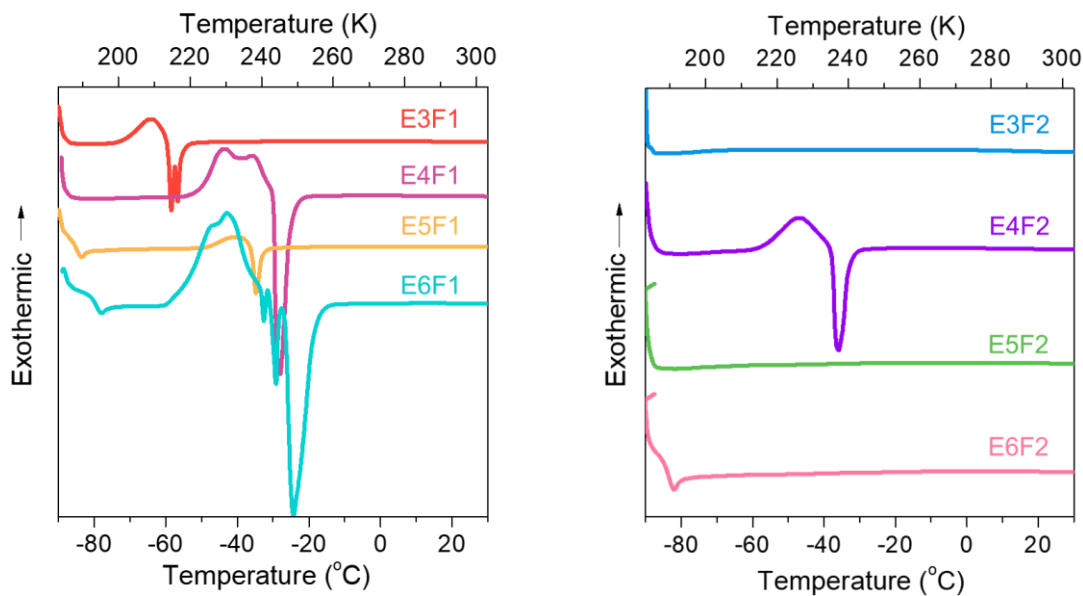


Figure S9. Differential scanning calorimetry (DSC) heating traces of pure solvents. All DSC tests were performed at heating or cooling rate of 10°C/min. The sample was first heated up to 80°C, and then looped twice between 80°C and –90°C. The heating traces of second loop are shown here because no transition was observed in cooling of all the F1 and F2 compounds and their LiFSA solutions.

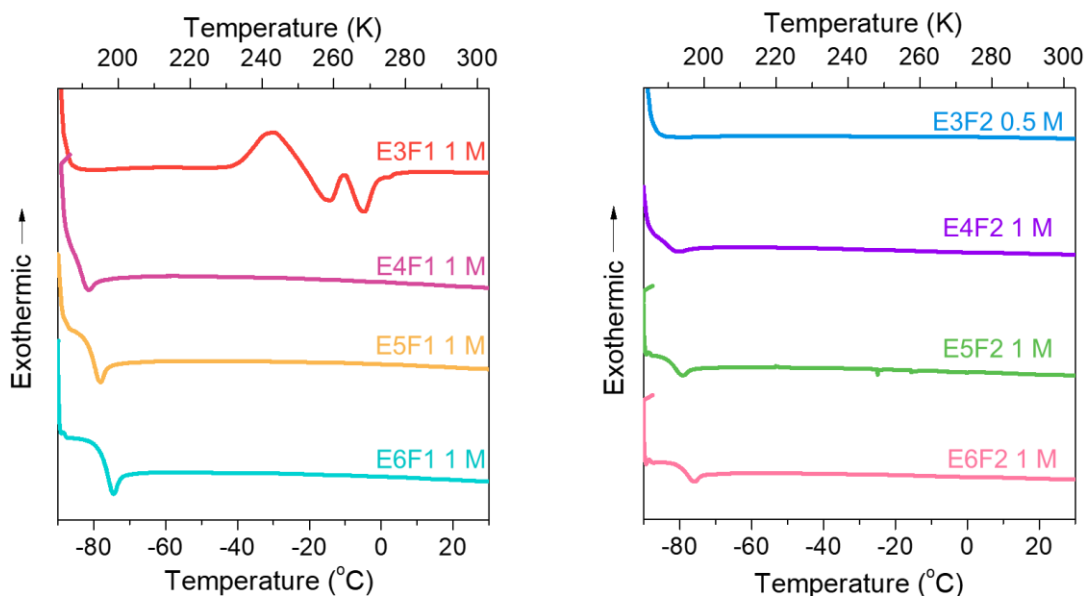


Figure S10. DSC heating traces of LiFSA in fluorinated ethers. All DSC tests were performed at heating or cooling rate of 10°C/min. The sample was first heated up to 80°C, and then looped twice between 80°C and –90°C. The heating traces of second loop are shown here because no transition was observed in cooling of all the F1 and F2 compounds and their LiFSA solutions.

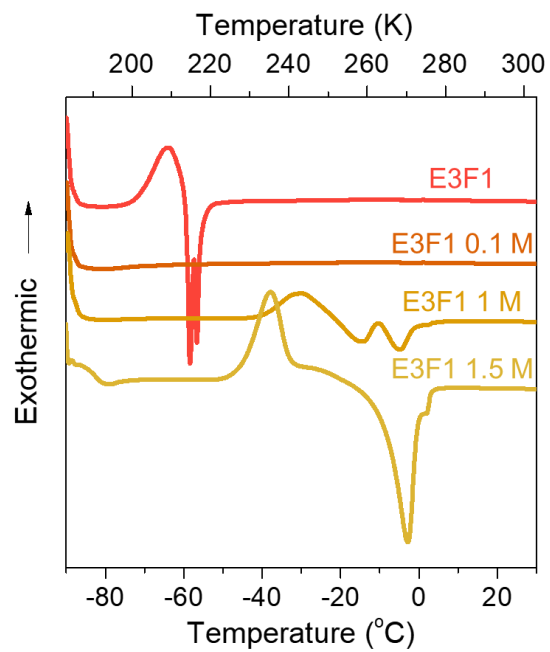


Figure S11. DSC of E3F1 and its LiFSA solutions. All DSC tests were performed at heating or cooling rate of 10°C/min. The sample was first heated up to 80°C, and then looped twice between 80°C and -90°C. The heating traces of second loop are shown here.

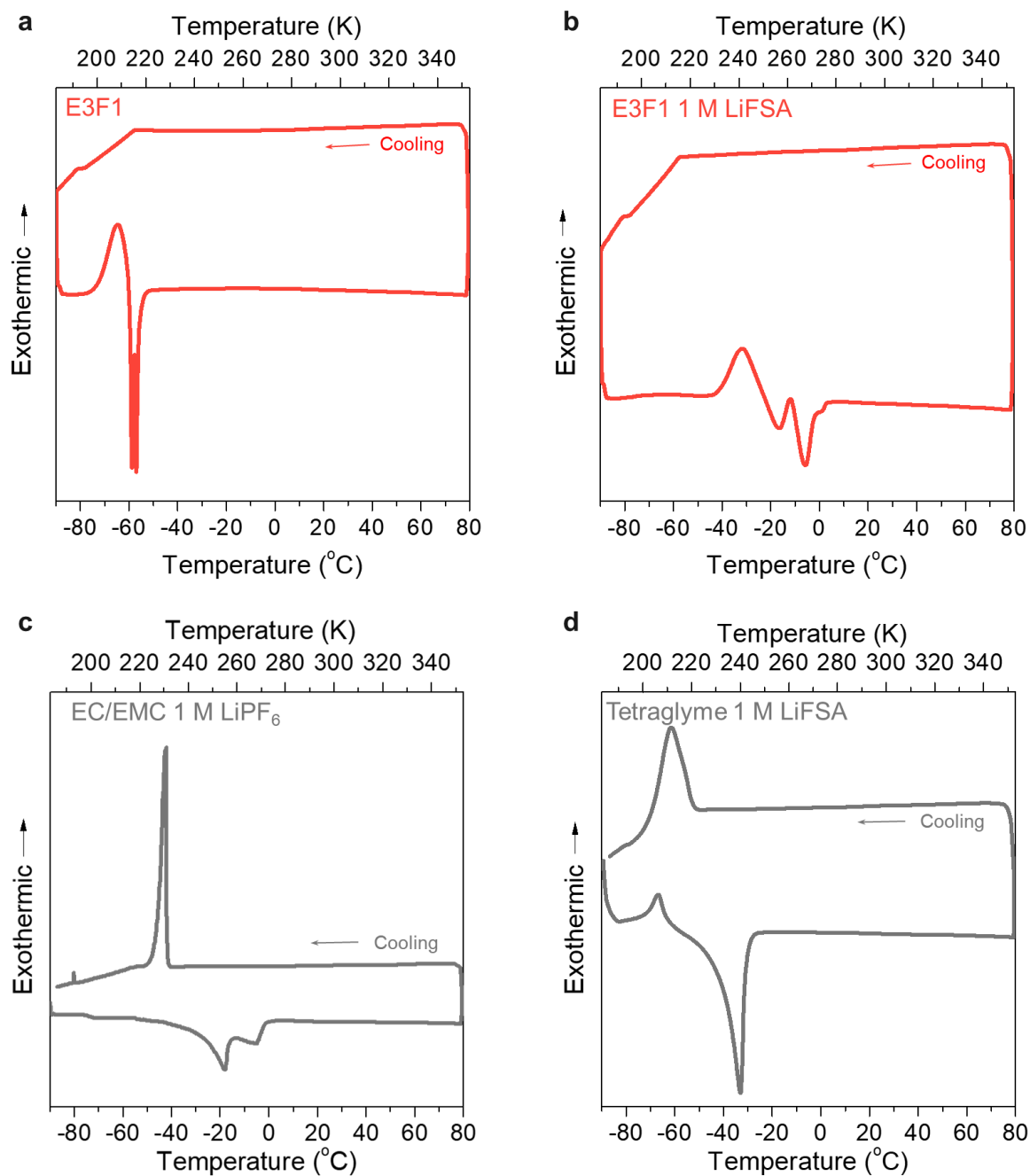


Figure S12. The second loop of DSC data for (a) E3F1, (b) 1 M LiFSA in E3F1, (c) commercially used carbonate electrolyte and (d) 1 M LiFSA in tetraglyme. In the second loop starting at 80°C, no transition was observed in cooling of all the pure F1 and F2 compounds and their LiFSA solutions under investigation. Here only E3F1 and its 1 M LiFSA solution are shown as examples. In contrast, the commercial carbonate electrolyte and 1 M LiFSA in tetraglyme electrolyte has strong crystallization peaks in the cooling trace, indicating their tendency to crystallize at low temperature. The differences in DSC shown here correlates to the low temperature conductivity data shown in Figure S6.

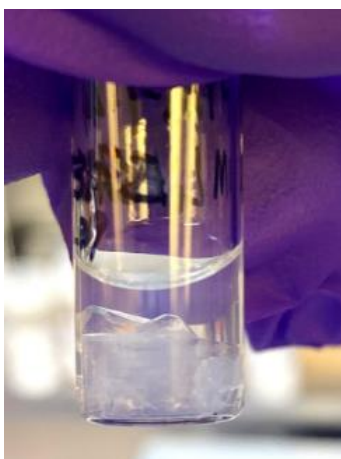


Figure S13. Digital photo of the 1 M LiFSA in E3F2 solution that crystallized at room temperature. Photo was taken using an Apple iPhone SE smartphone.

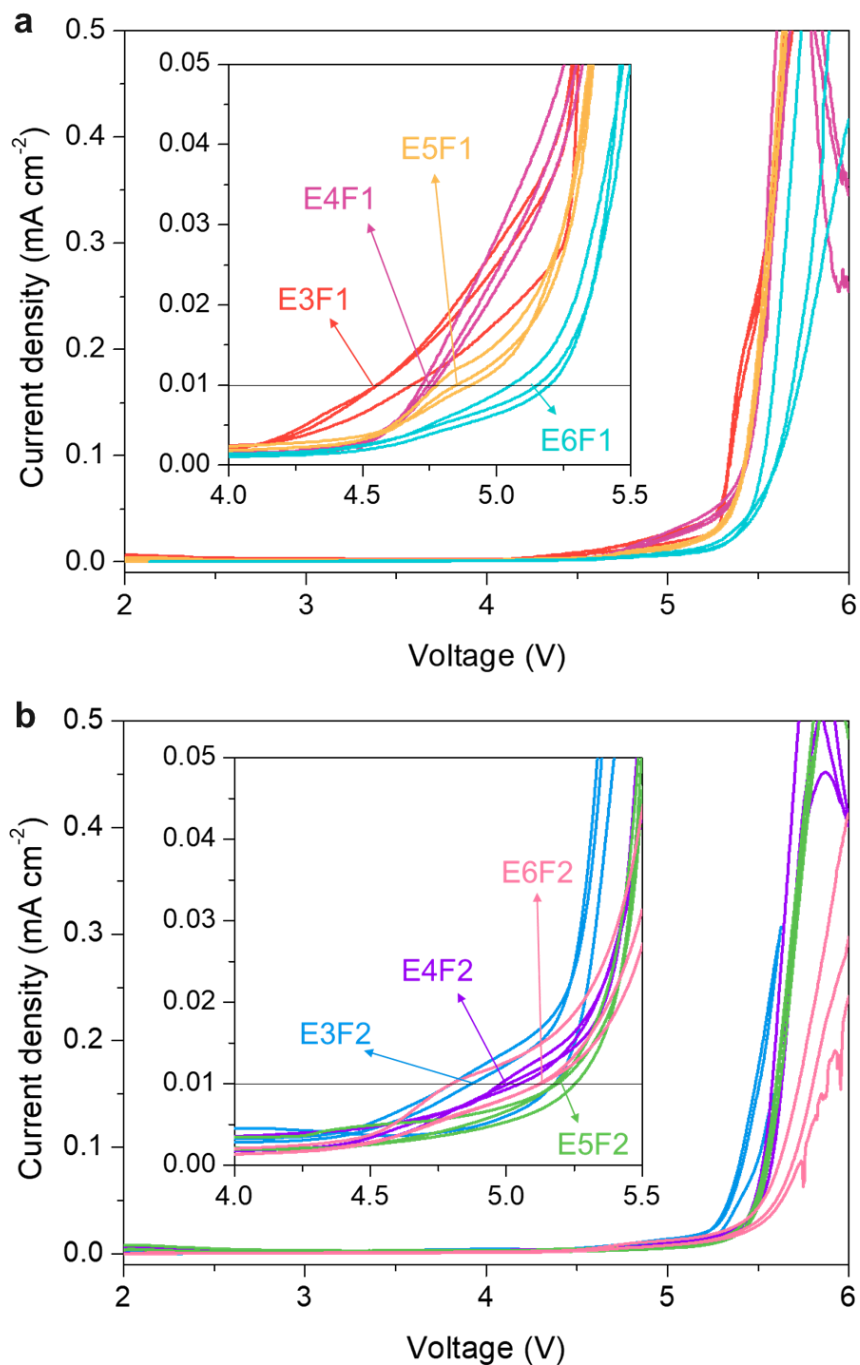


Figure S14. Linear sweep voltammetry (LSV) of 1 M LiFSA in (a) F1 and (b) F2 compounds using stainless steel as working electrode and lithium metal as the counter and reference. Three parallel experiments of each electrolyte are shown here for reproducibility. Each line represents a different coin cell.

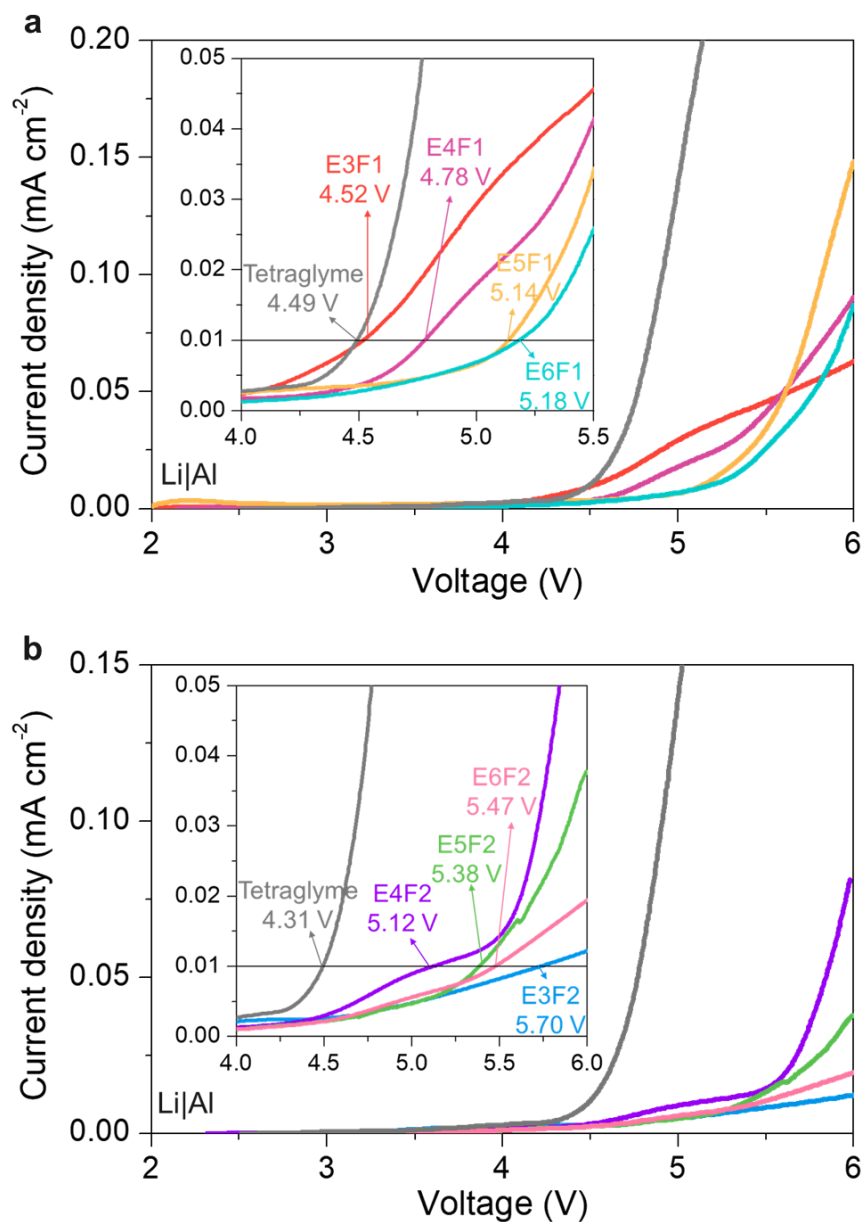


Figure S15. Linear sweep voltammetry (LSV) of 1 M LiFSA in F1 and F2 compounds with tetraglyme as control using aluminum as working electrode. The oxidative stability with Al working electrode follows the same trend as with SS electrode except for E3F2. The unexpected low current density of 1 M LiFSA in E3F2 is very likely due to crystallization (as shown in the digital photo in Figure S13).

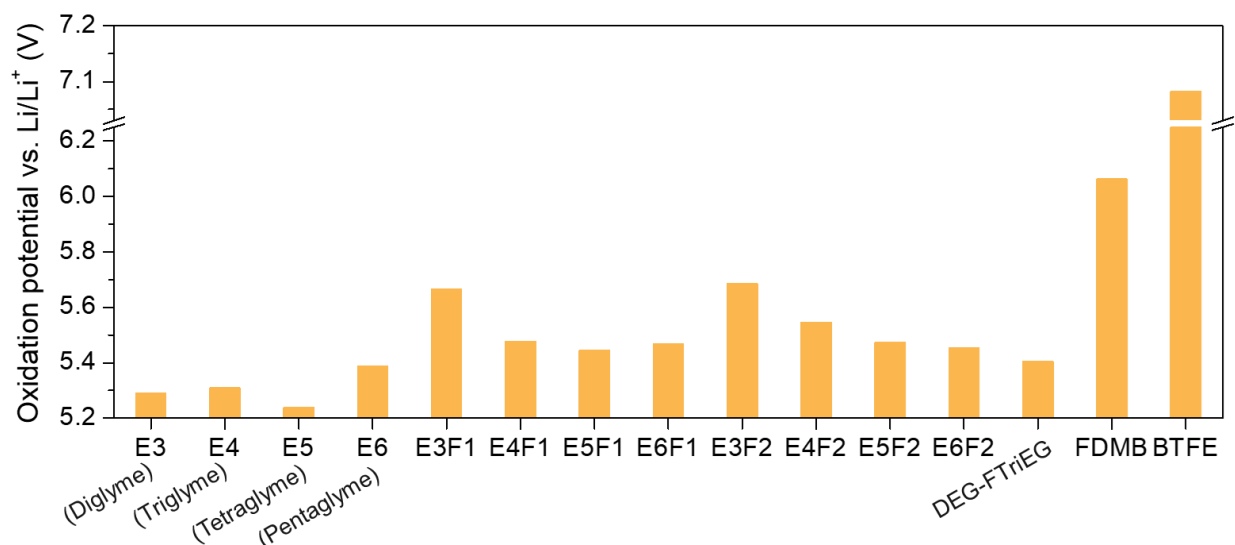


Figure S16. Adiabatic oxidation potential calculated by DFT. As a measure of oxidative stability, the oxidation potential data shows that fluoroether compounds have overall improved oxidative stability compared to traditional ethers (E3 to E6). In addition, oxidation potential decreases with increasing ether length in the F1 and F2 compounds, which agrees with intuition but contradicts the experimental oxidative stability trend as described in the main manuscript. DEG-FTriEG molecular structure can be found in Amanchukwu et al.² FDMB refers to fluorinated 1,4-dimethoxybutane from Yu et al.³ BTFE refers to Bis(2,2,2-trifluoroethyl) ether.

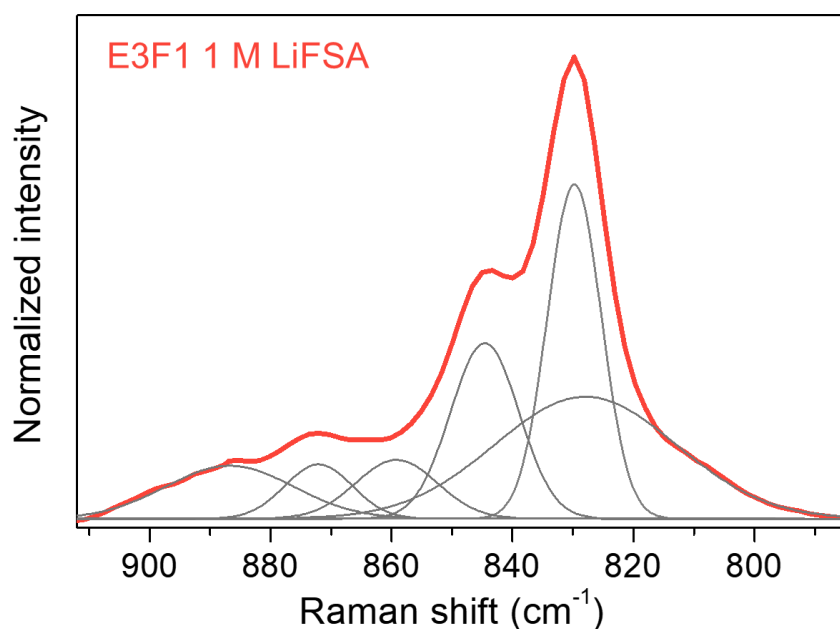


Figure S17. Raman spectra of 1 M LiFSA in E3F1. Six components were used to fit the range between 770 and 930 cm^{-1} . Fitting was done with Origin Pro 8.5 software.

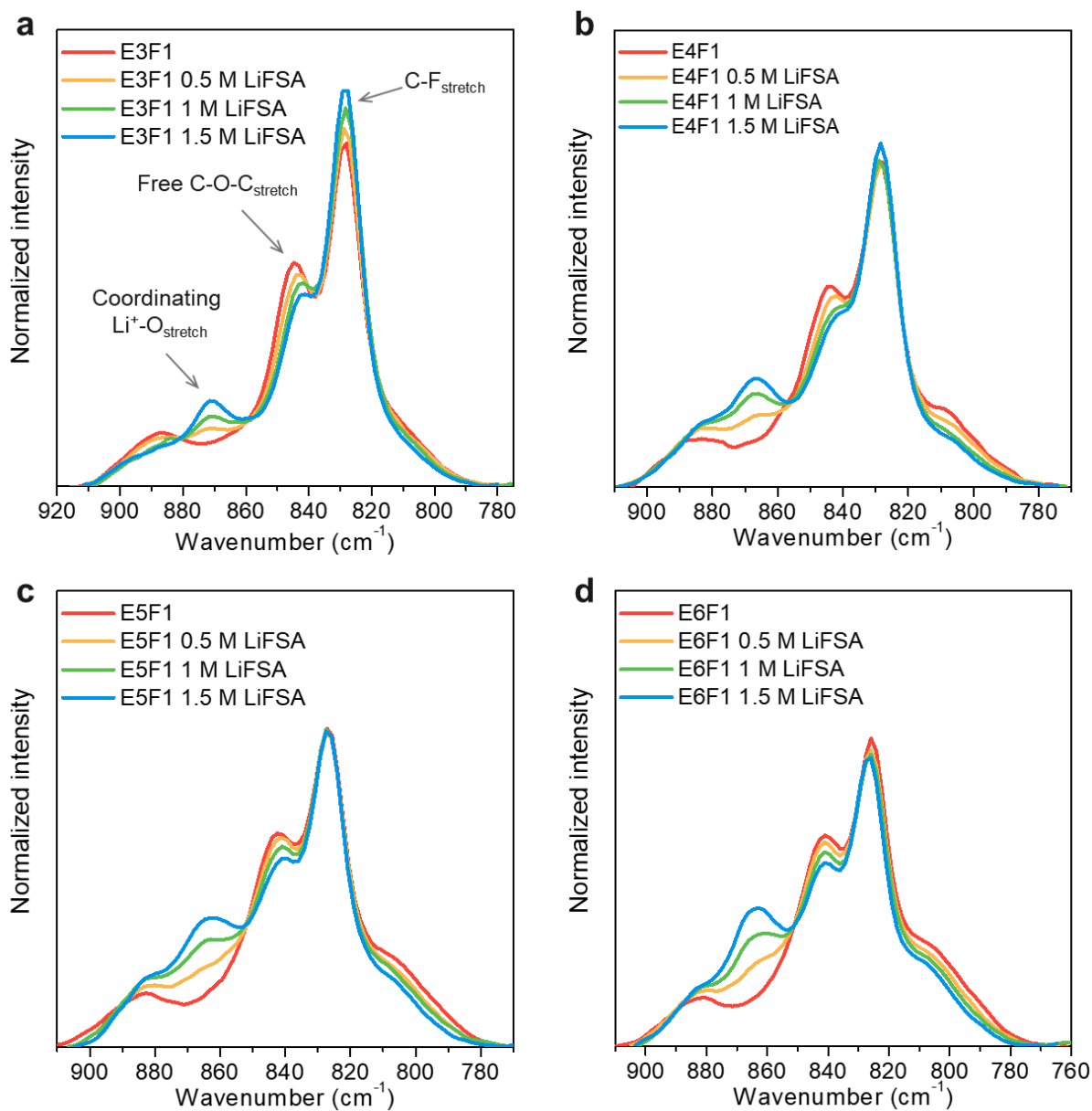


Figure S18. The variation of Raman spectra with LiFSA concentration in (a) E3F1, (b) E4F1, (c) E5F1 and (d) E6F1. The Raman spectra were normalized by their total area and aligned according to the C–F stretching peak ($\sim 828 \text{ cm}^{-1}$). As the LiFSA salt concentration increases from 0 to 1.5 M, the free C–O–C stretching peak around 843 cm^{-1} shrink significantly while the Li^+ –O breathing peak around 867 cm^{-1} increases. The vibration modes indicated in E3F1 correspond to the same functional groups in E4F1, E5F1, and E6F1.

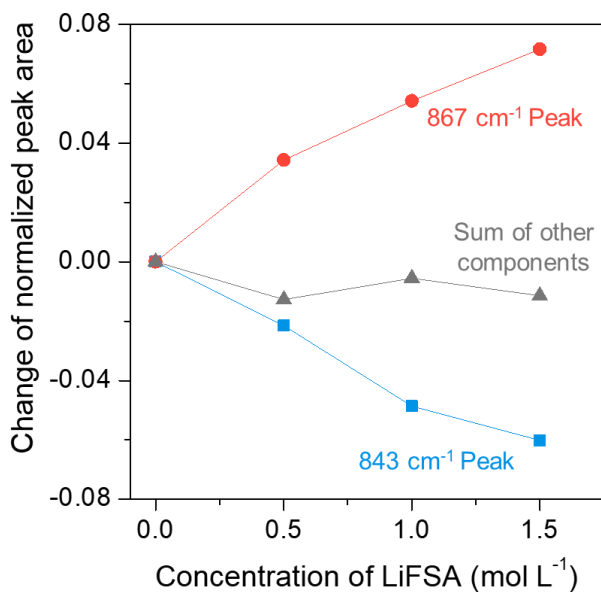


Figure S19. The evolution of Raman peak area with LiFSA concentration in E3F1. The peak area of every component shown in Figure S17 was first normalized to the total area and then subtracted by the normalized area value of pure E3F1 to calculate the change in normalized peak area. As shown here, the normalized peak area of 867 cm⁻¹ and 843 cm⁻¹ component varies significantly with salt concentration while the sum of others is almost constant.

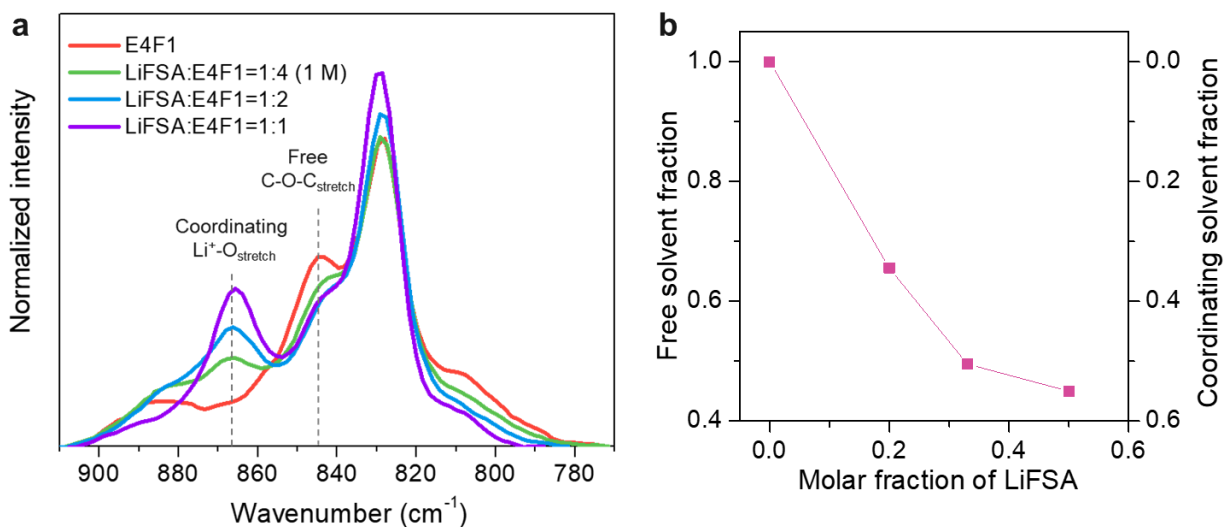


Figure S20. (a) The variation of Raman spectra with E4F1/LiFSA molar ratio. (b) The fraction of free and coordinating solvent as a function of LiFSA molar fraction. When the molar fraction of salt increase from 0 to 0.5 (pure E4F1 to LiFSA:E4F1 = 1:1), more solvent molecules participate in coordination, giving rise to lower free solvent fraction.

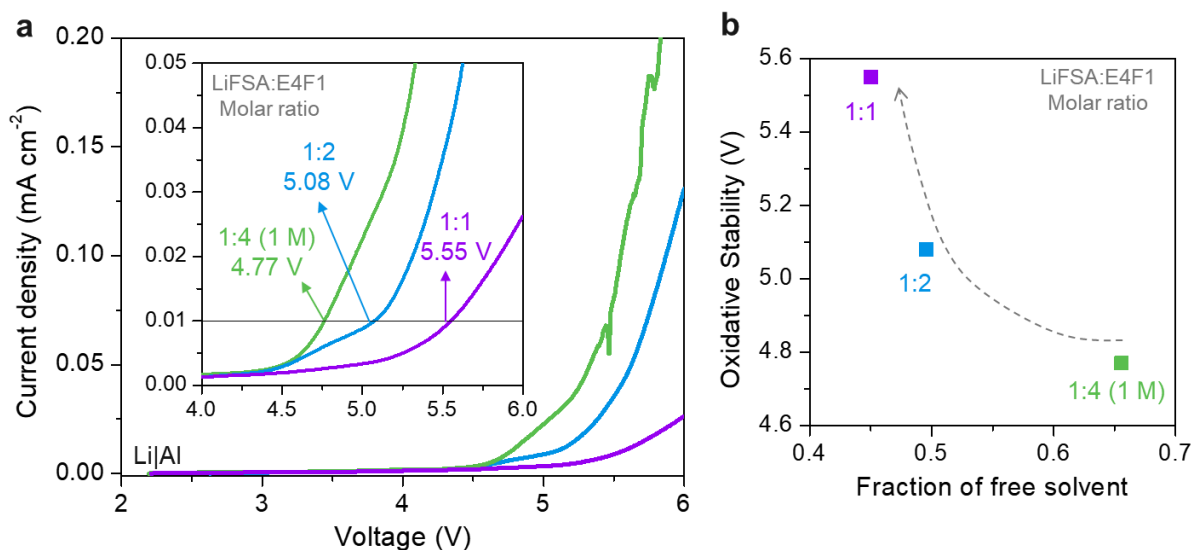


Figure S21. (a) The linear sweep voltammetry (LSV) of LiFSA/E4F1 solution in different molar ratios. (b) Oxidative stability of LiFSA/E4F1 electrolytes as a function of free solvent fraction. When the molar ratio LiFSA:E4F1 increases from 1:4 to 1:1, more solvent molecules participate in ion solvation and oxidative stability is enhanced by reduced free solvent fraction.

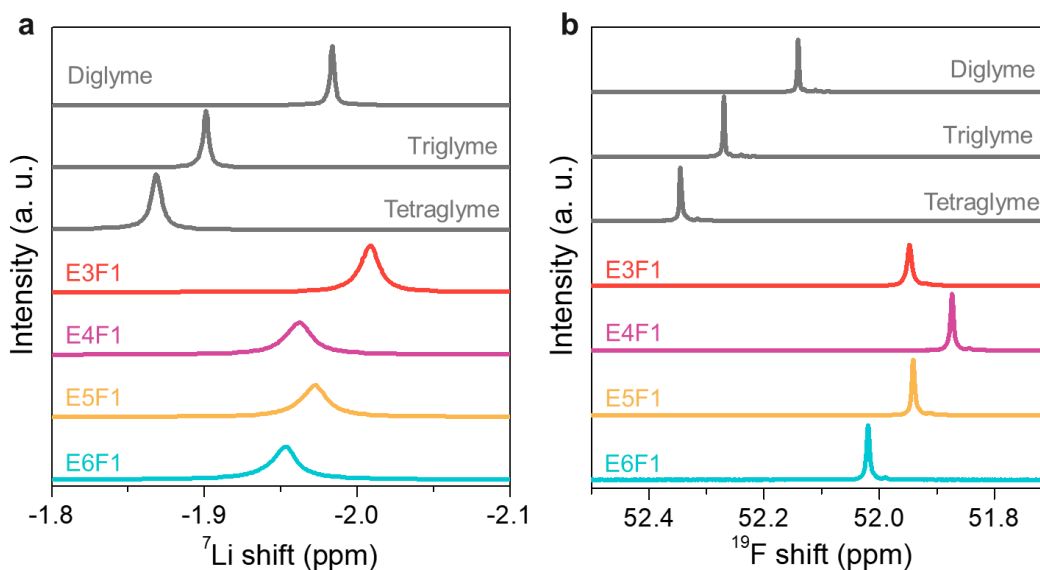


Figure S22. (a) ⁷Li and (b) ¹⁹F NMR shifts of 1 M LiFSA in F1 compounds and glymes as controls. ⁷Li shift was referenced to 0.1 M LiClO₄ in CD₃CN (-2.80 ppm)⁴ while ¹⁹F shift was referenced to trifluorotoluene (-62.5 ppm)⁵. The peaks of Li⁺ and FSA⁻ both overall shift to higher field with decreasing molecular size from tetraglyme to diglyme or from E6F1 to E4F1 except for E3F1. In ⁷Li NMR, E3F1 has an additional upfield shift of around 0.05 ppm from the other F1 electrolytes. In ¹⁹F NMR, E3F1 shifts to lower field from E4F1 as opposite to the trend. The distinct behavior of E3F1 can be explained by intensive ion pairing.

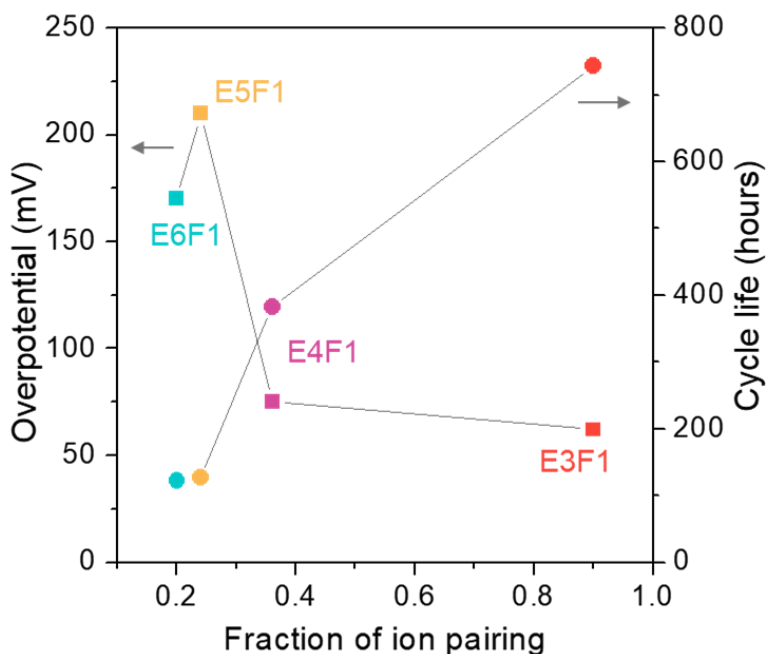


Figure S23. Summary of the Li/Li symmetric cells cycling results using 1 M LiFSA in F1 compounds as electrolytes. As the length of the fluorinated ether decreases from E6F1 to E3F1, weaker ion solvation (indicated by higher fraction of ion pairing) leads to lower overpotential and longer cycle life. The fraction of ion pairing was quantified from the fitting of FSA peak shown in Figure 4a: Fraction of ion pairing = (Area of AGG + Area of CIP)/Total area.

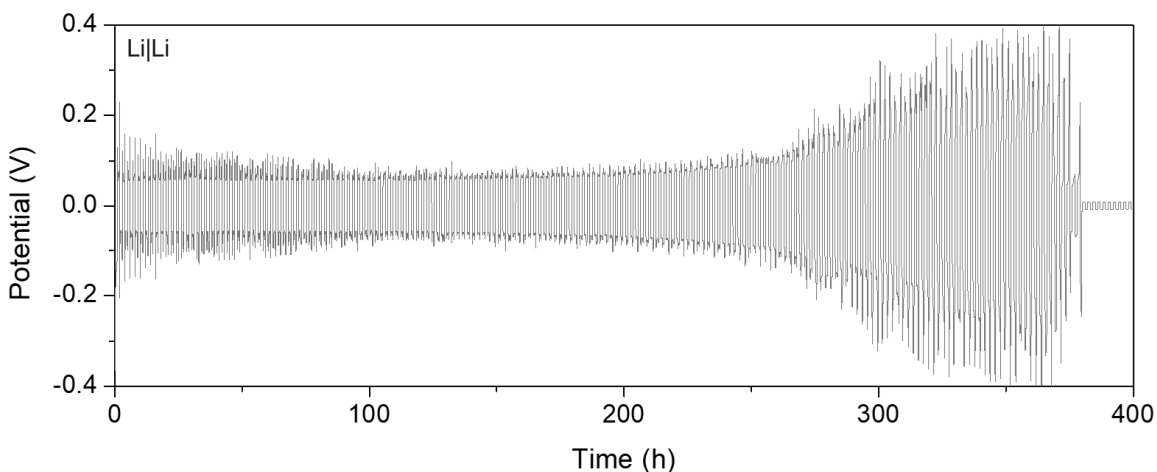


Figure S24. The voltage versus time plot of Li/Li symmetric cells using 1 M LiPF₆ in EC/DMC=50/50 (v/v). Celgard 3501 separator was used to replace Celgard 2325 for better wetting. The cells were cycled at a current density of 1 mA cm⁻² to 1 mA h cm⁻² after five formation cycles at 0.02 mA cm⁻² to 0.1 mA h cm⁻².

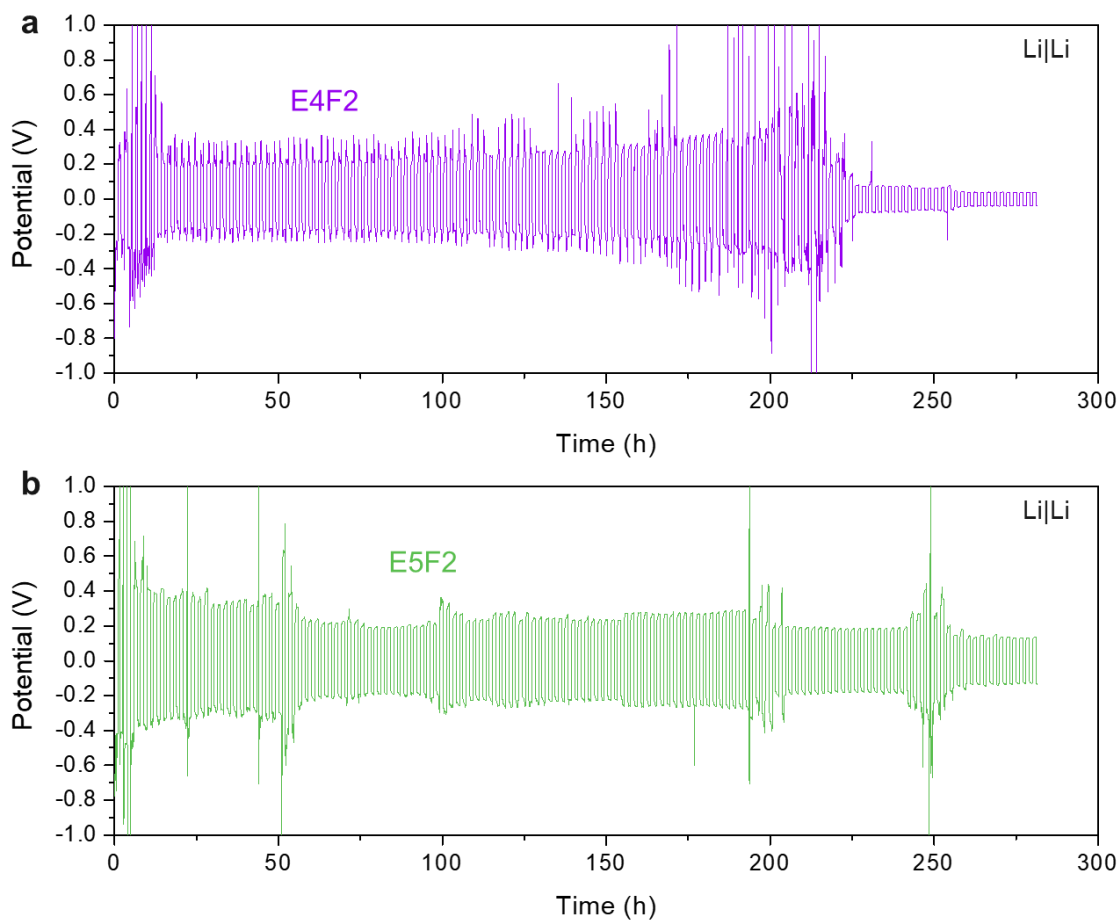


Figure S25. Voltage versus time plot of Li/Li symmetric cells using 1 M LiFSA in (a) E4F2 and (b) E5F2 as electrolytes. Cells were cycled at a current density of 1 mA cm^{-2} to 1 mA h cm^{-2} after five formation cycles at 0.02 mA cm^{-2} to 0.1 mA h cm^{-2} . The poor performance of E4F2 and E5F2 could be attributed to their inferior ionic conductivities.

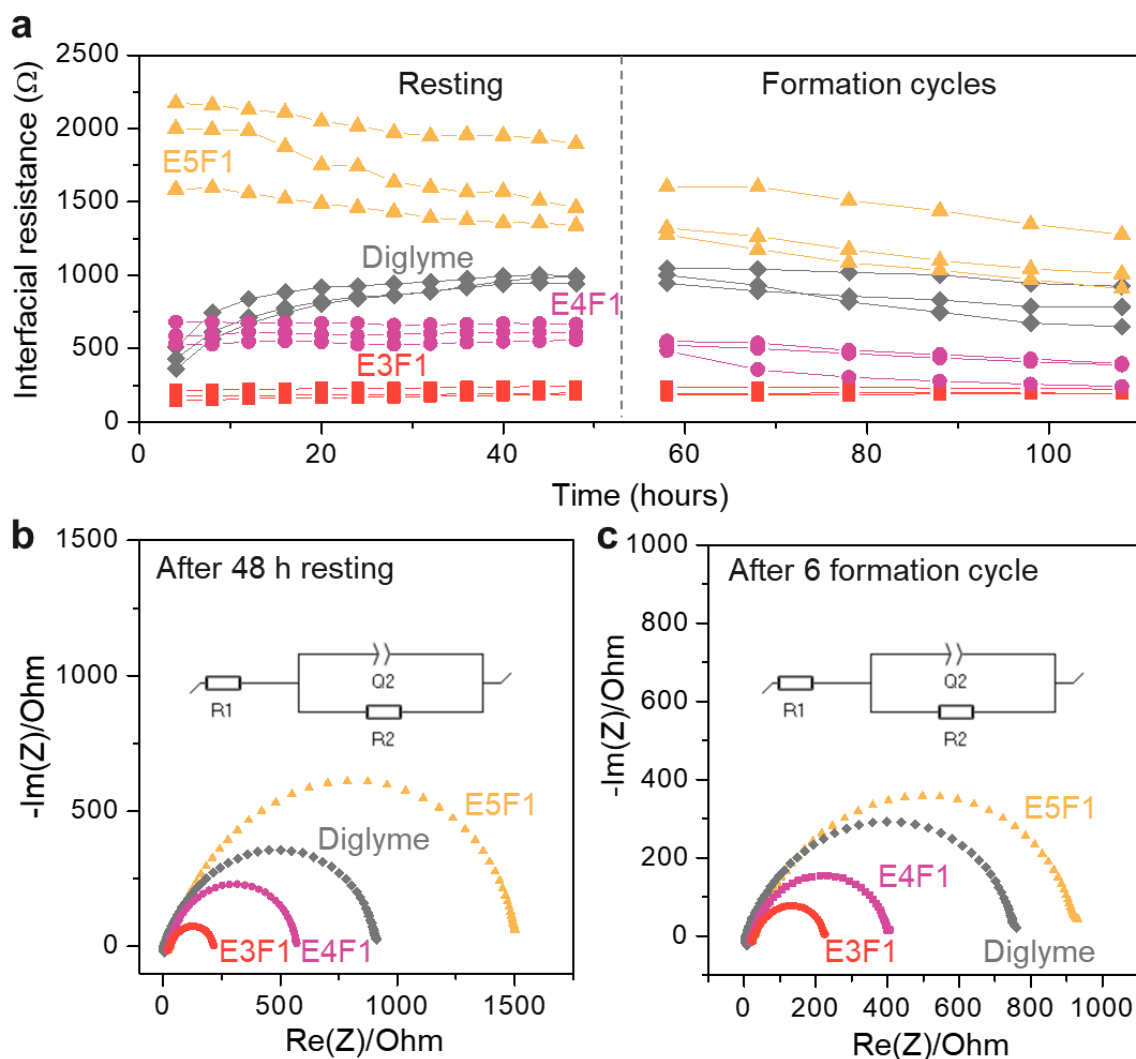


Figure S26. (a) The interfacial resistance of Li/Li symmetric cells using 1 M LiFSA in F1 compounds and diglyme as electrolytes obtained using EIS. Each line represents a different Li/Li cell. The Nyquist plot of electrochemical impedance spectra (b) after 48 hours resting and (c) after 6 formation cycles. Insets in (b) and (c): Equivalent circuit used for fitting. The value of R2 was taken as interfacial resistance. Fitting parameters of the E3F1 cell are shown below as an example.

	R1 (Ω)	R2 (Ω)	Q2 (F·s ^(a-1))	a2
E3F1 after 48 h resting	31.49	196.5	7.939e-6	0.8269
E3F1 after 6 formation cycle	30.54	192.7	8.004e-6	0.823

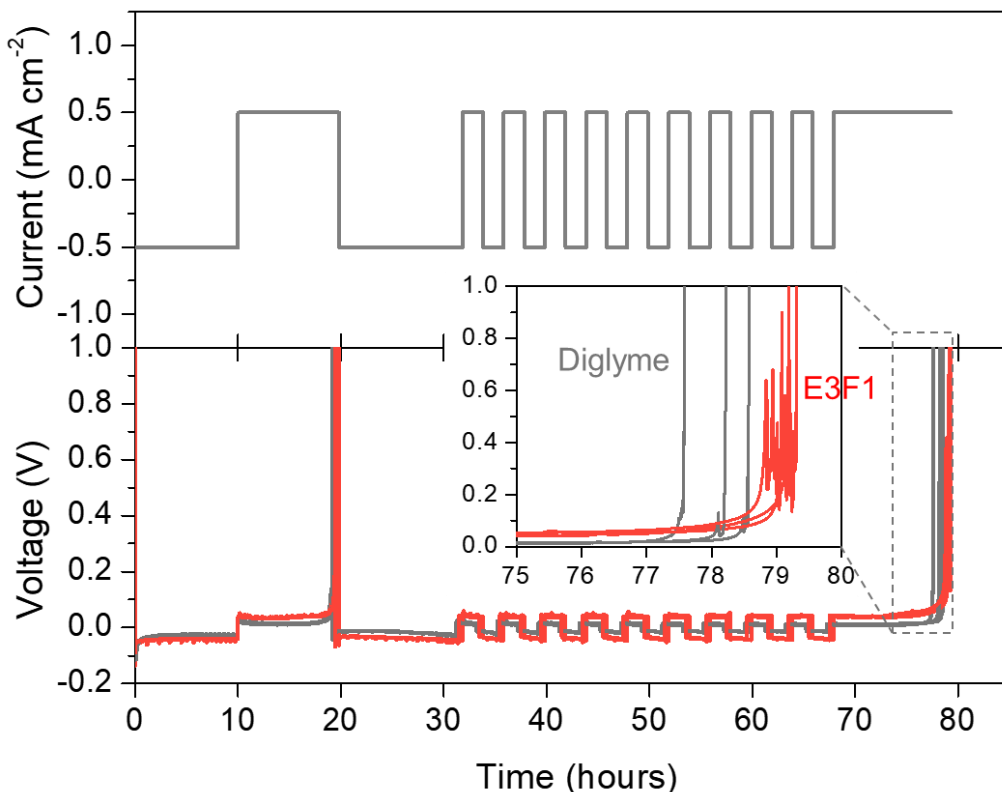


Figure S27. Coulombic efficiency (CE) measurements in Li/Cu cells using Aurbach protocol⁶ at a current density of 0.5 mA/cm². Inset: zoom-in view of the last stripping. 1 M LiFSA in E3F1 electrolyte has higher average CE than diglyme (98.9% vs 95.7%). CE data of each individual cell are shown below.

	Cell 1 (%)	Cell 2 (%)	Cell 3 (%)	Average CE (%)
E3F1	98.1	98.9	99.6	98.9
Diglyme	94.5	96.3	96.3	95.7

In this protocol, cells were first precycled by depositing lithium on copper electrode for 10 hours and then stripping to 1 V (the initial ~20 hours of the traces). Then, a ten-hour deposition was made (at 0.5 mA/cm²). Then, 10 deposition and stripping cycles at 0.5 mA/cm² to 1 mA h/cm² was performed. Finally, lithium was stripped from copper electrode until voltage reaches 1 V. CE was calculated following the equation below:

$$CE = \frac{\text{Total stripping capacity}}{\text{Total depositing capacity}} = \frac{\text{Capacity of the final stripping} + 10 \text{ mA h/cm}^2 (10 \text{ cycles})}{5 \text{ mA h/cm}^2 (\text{initial deposition}) + 10 \text{ mA h/cm}^2 (10 \text{ cycles})}$$

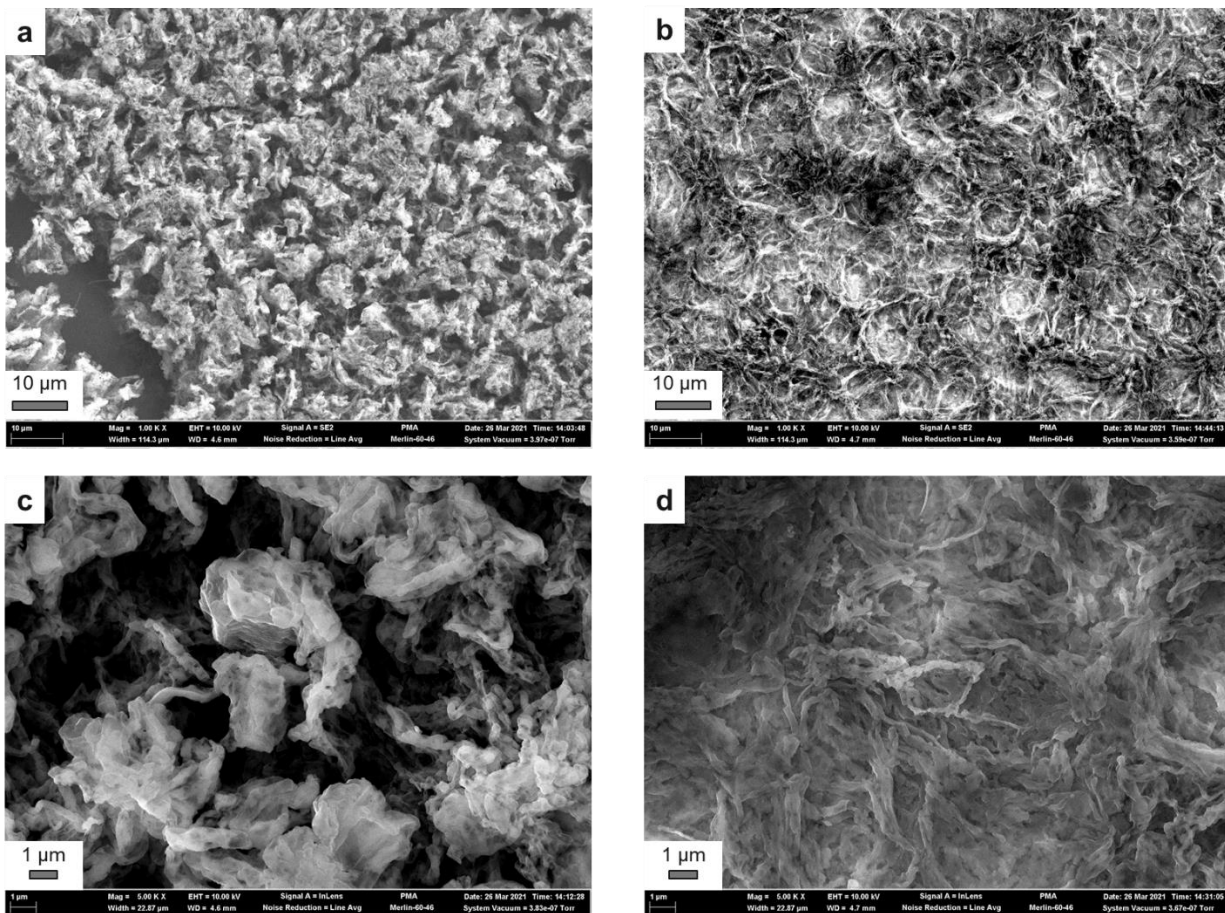


Figure S28. Scanning electron microscopy (SEM) images of residual lithium on copper electrode after one lithium deposition/stripping cycle using 1 M LiFSA in (a, c) diglyme and (b, d) E3F1. A smooth layer of dead lithium is observed in E3F1 while the morphology in diglyme is marked by strong contrast in height. The smoother morphology in E3F1 indicates less dead lithium formation, which agrees with the higher Coulombic efficiency observed on E3F1. To prepare the lithium sample, lithium was first deposited on copper electrode at a current rate of 1 mA/cm² to 1 mA h/cm² capacity and then stripped to a cut off voltage of 1 V.

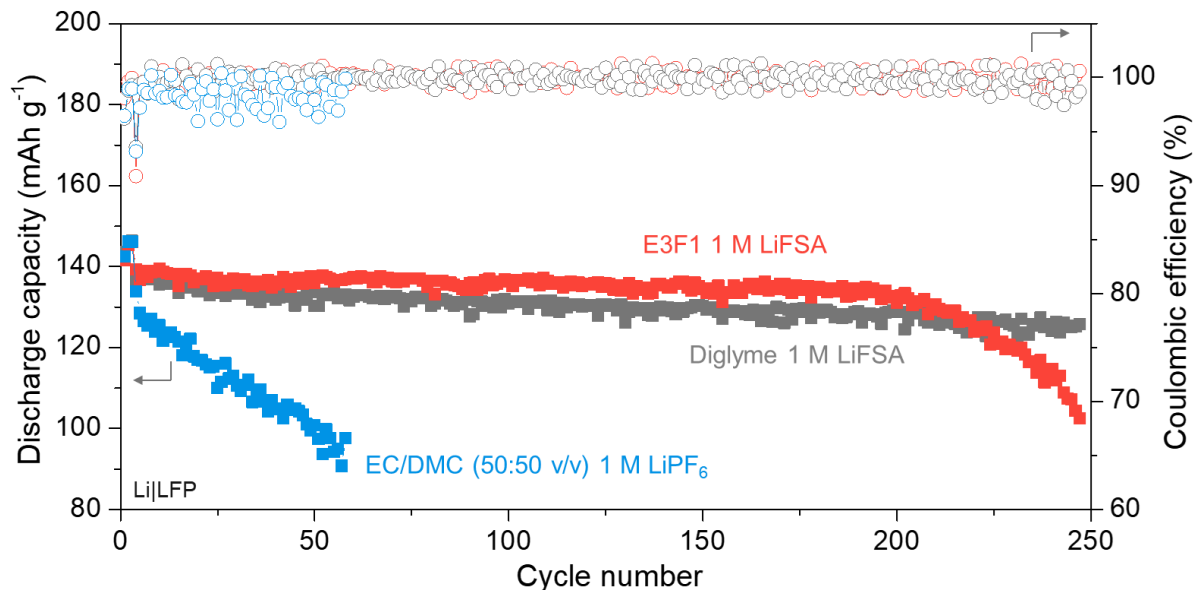


Figure S29. Galvanostatic cycling of Li/LFP cells at a current rate of C/3. Both diglyme and E3F1 maintain almost 100% Coulombic efficiency and stable discharge capacity for more than 200 cycles while the capacity of commercial carbonate electrolyte decays constantly. The cells were cycled between 2.9 V and 3.8 V at C/3 current density after three formation cycles at C/20.

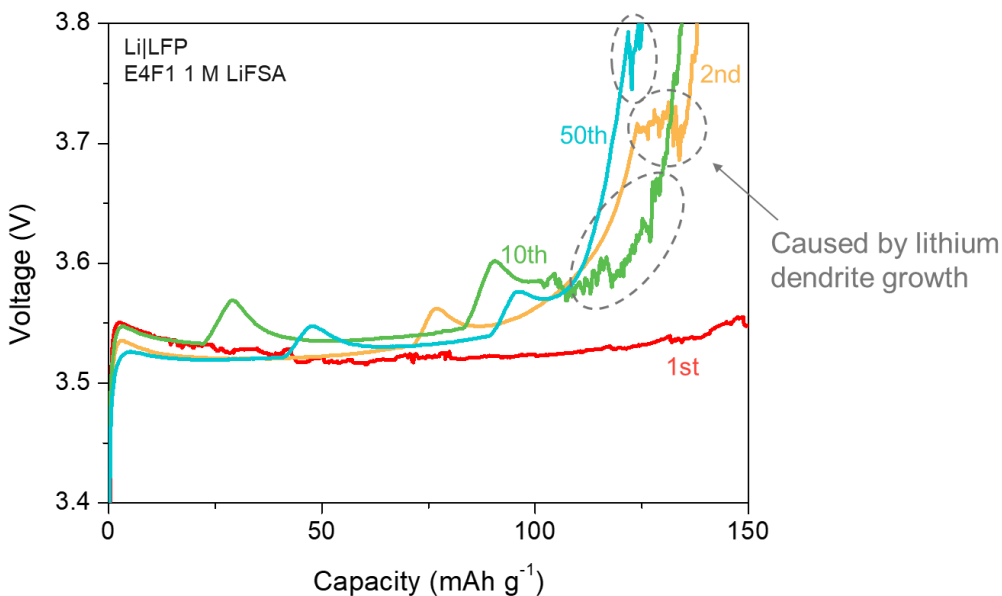


Figure S30. Charging profile of Li/LFP cell using 1 M LiFSA in E4F1 as electrolytes. The cell was cycled between 2.9 V and 3.8 V at C/3 current density after three formation cycles at C/20. The representative cycles shown here all have noisy voltage response, which has been reported as caused by lithium dendrite growth.⁷

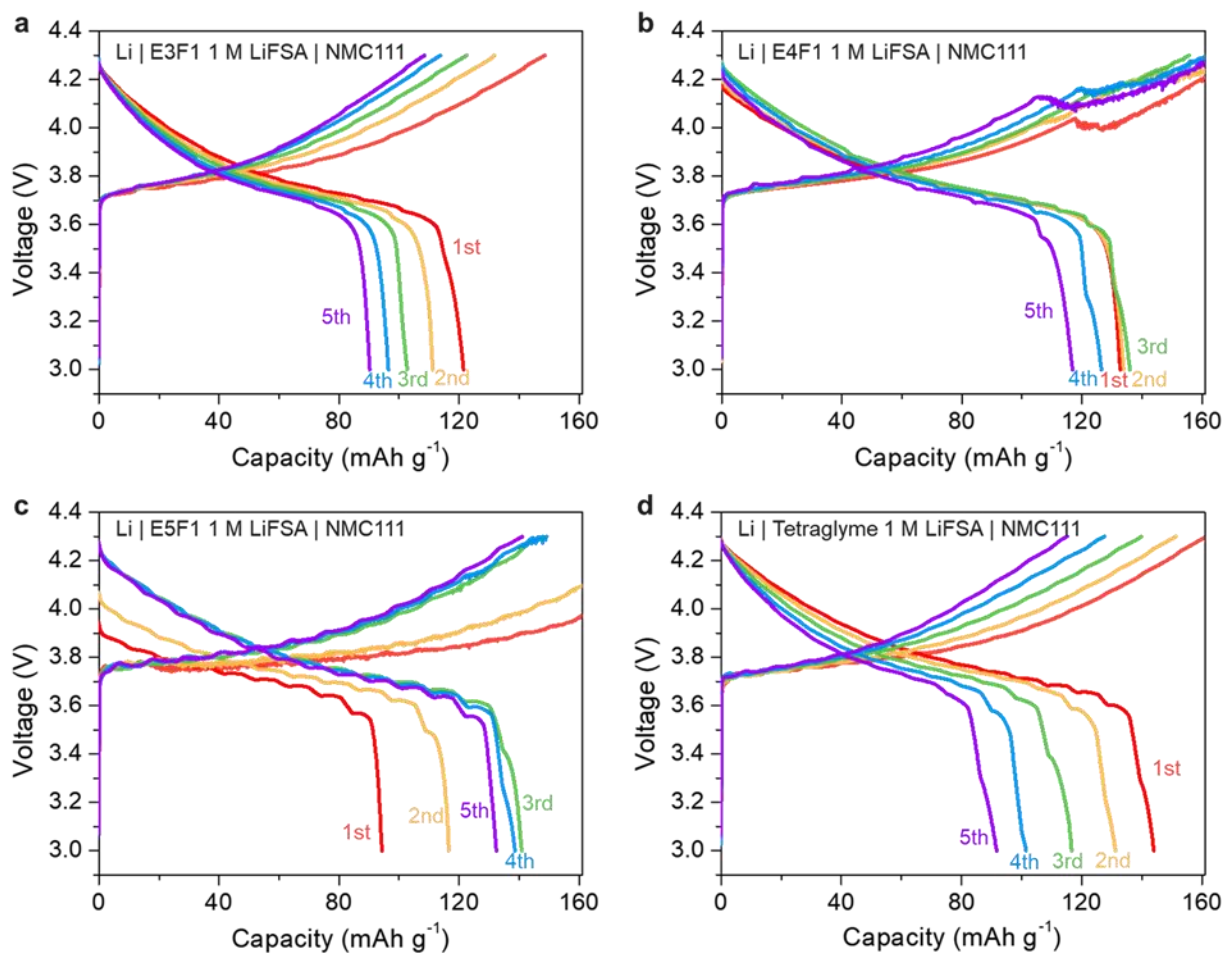


Figure S31. Galvanostatic cycling of Li/NMC111 cells at current rate of C/10 using 1 M LiFSA in (a) E3F1, (b) E4F1, (c) E5F1 and (d) tetraglyme as electrolytes. E4F1 and E5F1 cells are able to retain around 120 mA h/g discharge capacity in five cycles whereas the capacity of E3F1 and tetraglyme decay to 90 mA h/g at the fifth cycle. The better performance of E4F1 and E5F1 with 4.3 V class NMC 111 electrode reflects their higher oxidative stability compared to E3F1 and tetraglyme. However, noisy voltage response observed in E4F1 and E5F1 indicates lithium dendrites may still grow⁷, which leads to unsatisfactory Coulombic efficiency. Li/NMC 111 cells were cycled between 3–4.3 V. Three formation cycles at C/30 were performed prior to the test.

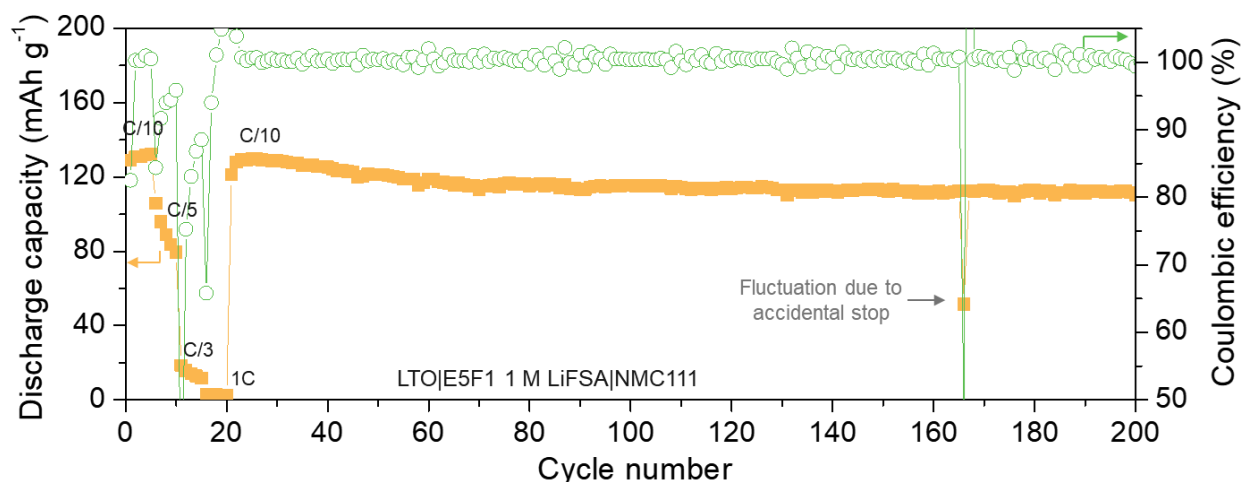


Figure S32. Galvanostatic cycling of LTO/NMC111 cells at current rates between C/10 and 1C using 1 M LiFSA in E5F1 electrolyte. At a slow rate of C/10, E5F1 electrolyte can be cycled repeatedly for more than 200 times with $\sim 86\%$ capacity retention. The fluctuation marked around 166th cycle was due to an accidental stop of the test.

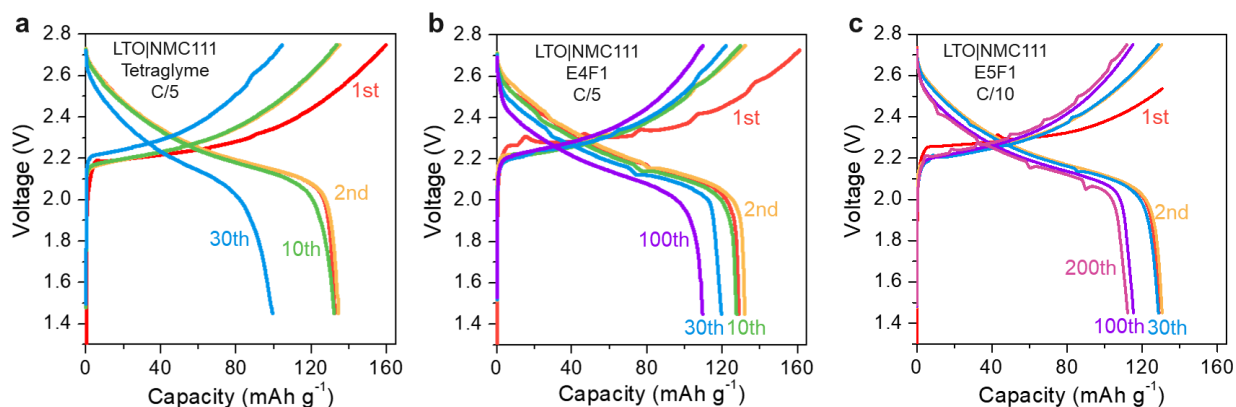


Figure S33. Galvanostatic cycling of LTO/NMC111 cells using (a) 1 M LiFSA in tetraglyme electrolyte, (b) 1 M LiFSA in E4F1 electrolyte, and (c) 1 M LiFSA in E5F1 electrolyte. Tetraglyme and E4F1 cells were cycled at a constant current rate of C/5 while the E5F1 cell was cycled mostly at C/10 (cycling program shown in Figure S32). In the initial two cycles, they all have discharge capacity around 130 mA h g^{-1} . Afterward, E4F1 cell and E5F1 cell maintain discharge capacity of $\sim 110 \text{ mA h g}^{-1}$ ($> 80\%$) at 100th cycle and 200th cycle, respectively. In contrast, the capacity of tetraglyme cell drops below 90 mA h g^{-1} at the 30th cycle. The better performance of E4F1 and E5F1 with 4.3 V class NMC 111 cathode reflects the improvement in oxidative stability. The fact that E4F1 and E5F1 electrolytes produce smooth voltage profiles in LTO/NMC 111 cells verified the effects of lithium dendrite growth on Li/NMC 111 cells.

Table S1. Summary of yield and physical properties

Compound name	Yield (%)	Boiling point at 2 kPa (°C)	Density (g cm⁻³)
E3F1	75	55	1.28
E4F1	60	90	1.25
E5F1	63	95	1.24
E6F1	66	-	1.21
E3F2	80	70	1.37
E4F2	72	-	1.34
E5F2	59	-	1.31
E6F2	65	-	1.29

Table S2. Summary of DSC data.

Sample	T _g (°C)	T _c (°C)	T _m (°C)	ΔH _c (J g ⁻¹)	ΔH _m (J g ⁻¹)
E3F1	-	-64.4	-56.7 (-58.4)	10.8	9.3
E4F1	-	-44.0 (-35.3)	-28.1	38.7	32.7
E5F1	-85.5	-40.0	-34.8	5.3	5.1
E6F1	-79.9	-43.0 (-47.7)	-24.4 (-29.3)	72.1	66.7
E3F2	-	-	-	-	-
E4F2	-	-47.0	-36.0	10.9	9.7
E5F2	-	-	-	-	-
E6F2	-83.8	-	-	-	-
E3F1 1 M LiFSA	-	-30.3	-4.7 (-14.6)	7.0	7.8
E4F1 1 M LiFSA	-84.4	-	-	-	-
E5F1 1 M LiFSA	-81.0	-	-	-	-
E6F1 1 M LiFSA	-77.5	-	-	-	-
E4F2 1 M LiFSA	-83.3	-	-	-	-
E5F2 1 M LiFSA	-82.1	-	-	-	-
E6F2 1 M LiFSA	-78.9	-	-	-	-
Diglyme	-	-	-	-	-
Triglyme	-	-59.6*	-41.5	115.4	127.7
Tetraglyme	-	-34.9*	-25.9	97.5	105.1
Diglyme 1 M LiFSA	-	-	-	-	-
Triglyme 1 M LiFSA	-	-72.8 (-63.7)	-48.1	70.0	69.8
Tetraglyme 1 M LiFSA	-	-61.8* (-66.9)	-33.2	41.5	45.5
EC/DMC 1 M LiPF ₆	-	-42.5*	-18.2 (-5.9)	1.29	1.77

The transition points marked with “*” were observed in the cooling trace. For those samples having multiple crystallization/melting peaks, the temperature of second strongest peak was shown in bracket. The crystallization/melting enthalpy was calculated from the total area of corresponding peaks.

Table S3. Adiabatic redox energy data reproduced from reference paper

Compound name	Adiabatic oxidation energy (eV)		Adiabatic reduction energy (eV)	
	Reproduced	Reference ⁸	Reproduced	Reference ⁸
DMC	-8.23	-8.51	-1.42	-1.42
PC	-8.38	-8.38	-1.54	-1.54
THF	-6.93	-6.94	-0.17	-0.17
DMSO	-6.46	-6.46	-1.93	-1.92
Pyridine	-6.96	-6.94	-1.74	-1.74

The data shown in this table were calculated rigorously following the procedure of the stated reference paper.⁸ Most of the reproduced redox energy values matched well with reference. However, a significant difference was observed in the adiabatic oxidation energy of DMC, which may be due to different initial structure selection.

References

- (1) Saito, M.; Yamada, S.; Ishikawa, T.; Otsuka, H.; Ito, K.; Kubo, Y. Factors Influencing Fast Ion Transport in Glyme-Based Electrolytes for Rechargeable Lithium–Air Batteries. *RSC Adv.* **2017**, *7* (77), 49031–49040. <https://doi.org/10.1039/C7RA07501D>.
- (2) Amanchukwu, C. V.; Yu, Z.; Kong, X.; Qin, J.; Cui, Y.; Bao, Z. A New Class of Ionically Conducting Fluorinated Ether Electrolytes with High Electrochemical Stability. *J. Am. Chem. Soc.* **2020**, *142* (16), 7393–7403. <https://doi.org/10.1021/jacs.9b11056>.
- (3) Yu, Z.; Wang, H.; Kong, X.; Huang, W.; Tsao, Y.; Mackanic, D. G.; Wang, K.; Wang, X.; Huang, W.; Choudhury, S.; Zheng, Y.; Amanchukwu, C. V.; Hung, S. T.; Ma, Y.; Lomeli, E. G.; Qin, J.; Cui, Y.; Bao, Z. Molecular Design for Electrolyte Solvents Enabling Energy-Dense and Long-Cycling Lithium Metal Batteries. *Nat. Energy* **2020**, *5*, 526–533. <https://doi.org/10.1038/s41560-020-0634-5>.
- (4) Günther, H. Lithium NMR. In *eMagRes*; 2007. <https://doi.org/https://doi.org/10.1002/9780470034590.emrstm0273>.
- (5) Li, C.; Suzuki, K.; Yamaguchi, K.; Mizuno, N. Phosphovanadomolybdic Acid Catalyzed Direct C–H Trifluoromethylation of (Hetero)Arenes Using NaSO₂CF₃ as the CF₃ Source and O₂ as the Terminal Oxidant. *New J. Chem.* **2017**, *41* (4), 1417–1420. <https://doi.org/10.1039/C6NJ03654F>.
- (6) Adams, B. D.; Zheng, J.; Ren, X.; Xu, W.; Zhang, J.-G. Accurate Determination of Coulombic Efficiency for Lithium Metal Anodes and Lithium Metal Batteries. *Adv. Energy Mater.* **2018**, *8* (7), 1702097. <https://doi.org/https://doi.org/10.1002/aenm.201702097>.
- (7) Homann, G.; Stolz, L.; Nair, J.; Laskovic, I. C.; Winter, M.; Kasnatscheew, J. Poly(Ethylene Oxide)-Based Electrolyte for Solid-State-Lithium-Batteries with High Voltage Positive Electrodes: Evaluating the Role of Electrolyte Oxidation in Rapid Cell Failure. *Sci. Rep.* **2020**, *10* (1), 4390. <https://doi.org/10.1038/s41598-020-61373-9>.
- (8) Feng, S.; Chen, M.; Giordano, L.; Huang, M.; Zhang, W.; Amanchukwu, C. V.; Anandakathir, R.; Shao-Horn, Y.; Johnson, J. A. Mapping a Stable Solvent Structure Landscape for Aprotic Li–Air Battery Organic Electrolytes. *J. Mater. Chem. A* **2017**, *5* (45), 23987–23998. <https://doi.org/10.1039/C7TA08321A>.